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# Kinetic Analysis of Non-Isothermal Reduction of Silica-Supported Nickel Catalyst Precursors in a Hydrogen Atmosphere

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## Abstract

A series of silica-supported nickel catalyst precursors was synthesized with different SiO<sub>2</sub>/Ni molar ratios. Reduction of Ni catalyst precursors with different SiO<sub>2</sub>/Ni molar ratios under hydrogen atmosphere was investigated at different heating rates. Kinetic parameters were determined using Kissinger-Akahira-Sunose isoconversional and invariant kinetic parameters methods. It was found, that for all molar ratios, apparent activation energy (E<sub>a</sub>) is practically constant in conversion range of  $0.20 \le \alpha \le 0.80$ . In considered conversion range, following values of E<sub>a</sub> were found:  $134.5 \text{ kJmol}^{-1}$  (SiO<sub>2</sub>/Ni = 0.20),  $139.6 \text{ kJmol}^{-1}$  (SiO<sub>2</sub>/Ni = 0.80) and  $128.3 \text{ kJmol}^{-1}$  (SiO<sub>2</sub>/Ni molar ratios is a complex process and can be described by Šesták-Berggren autocatalytic model. It was found that the reaction is more Langmuir-Hinshelwood type, as hydrogen dissociates rapidly on surface nuclei and dissociated hydrogen reacts with Ni-O active system. It was concluded that reduction process proceeds through bulk nucleation which is dominant mechanism, where three-dimensional growth of crystals with polyhedron-like

morphology exist. It was found that Ni/Si ratio decrease after the reduction process. This has been explained by low Ni and higher Si surface concentrations. It has been disclosed that Ni dispersion decreases.

KEYWORDS: Autocatalysis; Catalyst support; Complexity; Kinetics; Reduction

#### INTRODUCTION

Catalysis plays a pivotal role in the physical and biological sciences, in the industrial production of chemicals, 90% of which are obtained via catalytic conversions (**Thomas and Thomas, 1997**). Catalysis research covers both multidisciplinary science and engineering.

In the field of heterogeneous catalysis, thermal analysis is used as a tool for investigating the influence of composition, preparation method and pre-treatment on the reactivity of catalyst surface or bulk with gases. Although thermal analysis is most frequently applied in the decomposition of solid materials, it also constitutes a suitable methodology to assess the kinetics of other thermally stimulated processes (**Brown, 2001**). Among thermo-analytical techniques, temperature-programmed desorption (TPD) and temperature-programmed reduction (TPR) are the tools that are most commonly used for characterizing heterogeneous catalysts. The temperature-programmed reduction (TPR) was inspired by the TPD technique and proposed in its present form by Robertson et al. (**Robertson et al., 1975**). In TPR, oxidic catalyst precursor is submitted to a programmed temperature rise (usually obeying the linear law represented as  $T = T_0 + \beta t$ , where *T* is the

absolute temperature,  $T_o$  is the initial temperature,  $\beta$  is the temperature increase rate (linear heating rate) and *t* is the time) under the flow of reducing gas mixture, and the consumption of the reducing agent is continuously monitored. Alongside catalyst characterization, temperature programmed techniques can be applied to mimic pretreatment procedures related to the operation of catalytic processes (**Kapteijn et al.**, **1999**). The application of TPD, TPR and related techniques was reviewed by Cvetanović and Amenomiya (**Cvetanović and Amenomiya**, **1972**), Falconer and Schwarz (**Falconer and Schwarz**, **1983**), Hurst et al. (**Hurst et al.**, **1982**), Lemaitre (**Lemaitre**, **1984**) and Bhatia et al. (**Bhatia et al.**, **1990**).

Generally, temperature programming is better than flash methods in the fact that the time and temperature are interrelated in programming techniques. One can, therefore, deduce suitable kinetic expressions for a desorption process or for a process taking place as a result of temperature programming. In essence, temperature programming appears to provide a handle to discriminate various individual steps involved in desorption, reduction or oxidation processes while isothermal methods can provide information on the overall process that takes place.

Silica supported nickel catalysts are used extensively in many industrial processes such as hydrogenation of edible oils and aromatics in distillate fuels. As all these catalysts require optimized physical and chemical properties, a great deal of research has been reported with the aim of developing controlled catalyst preparation procedures. Continuous efforts are made and literature presents numerous works relating preparation methods, catalysts structure and catalytic behavior (González-Marcos et al., 1997; Gil et al., 1994; Che et al., 1995; Stevenson et al., 1989).

It was reported previously (**Jovanović et al., 1998**) that good hydrogenation catalysts could be obtained if Ni(NO<sub>3</sub>)<sub>2</sub> is precipitated with Na<sub>2</sub>CO<sub>3</sub> onto a SiO<sub>2</sub> support (activated diatomite), with subsequent reduction in a hydrogen atmosphere. The possibilities for the characterization of the reduced catalysts are limited by their pyrophoric properties. Hence, after reduction, the catalyst must be passivated before any further manipulation. On the other hand, the precursor properties could be significantly changed during the reduction process, and the properties of the final catalyst are strongly dependent on the reduction conditions. More generally, the properties of the final catalyst, such as catalytic activity, selectivity etc., are strongly dependent on various process parameters used in the preparation treatment (**Nele et al., 1999; Bhering et al., 2002; Fierro and Peña, 2005**).

Bhering et al. (**Bhering et al., 2002**) used a statistical experimental setup to study the reduction step in the preparation of silica-supported nickel catalysts. These authors differentiated between precursors with weak nickel-support interactions and ones with strong nickel-support interactions. The strong metal support interactions were observed in a SiO<sub>2</sub>/Ni catalyst prepared via sol-gel synthesis (**Ueckert et al., 1997; Gronchi et al., 2003**). The weak interaction samples seem to be completely reduced even at low temperatures, while the strong interaction samples are reduced to different degrees at different temperatures due to high temperature stability of nickel silicate. In both cases, the specific area of the final catalyst depends strongly on the flow rate and hydrogen

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concentration. The formation of silicates should be avoided when highly reducible Ni precursors are supported. Therefore, low surface area silica supports and fast precipitation procedures (**Butler et al., 1996**) that result in supported basic nickel carbonate should be used. In addition, the mechanism of supported basic nickel carbonate reduction is not completely clear, and it is complicated by the fact that the reduction occurs in the temperature region corresponding to thermal decomposition of basic nickel carbonate. The kinetics of the reduction process could therefore give a new insight into the reaction mechanism of the investigated process and might lead to improvements of the catalysts (**L'vov and Galwey, 2012**).

The aim of this work is to perform kinetic analysis on non-isothermal reduction results for a series of silica-supported Ni-catalyst precursors, with different SiO<sub>2</sub>/Ni molar ratios (0.20, 0.80 and 1.15). The reduction of the investigated samples was studied by thermogravimetric analysis (TGA) with hydrogen as the reducing agent. Different kinetic models were tested against the experimental data and the kinetic parameters were estimated using different methods, as the integral isoconversional method (**Kissinger**, **1957; Akahira and Sunose, 1971**), the invariant kinetic parameters (IKP) method (**Lesnikovich and Levchik, 1983, 1985**) and the kinetic method introduced by Málek's (**Málek, 1992**). Also, in this paper, we will make a comparison of the results obtained with SiO<sub>2</sub>-supported and unsupported nickel catalysts during reduction process (**Martin et al., 1981; Jackson et al., 1999; Weatherbee and Bartholomew, 1984; Janković et al., 2008**). Also, we will discuss the possible contradictions in reduction kinetics of silicasupported and unsupported nickel catalysts (**Zieliński, 1995; Geus, 1988**).

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#### **EXPERIMENTAL PROCEDURE**

### Materials And Methods

A series of nickel catalyst precursors with different SiO<sub>2</sub>/Ni molar ratio values (0.20, 0.80 and 1.15) was synthesized according to a literature procedure (**Jovanović et al., 1998**). Nickel ions and promoter ions (Mg) where precipitated at an elevated temperature (90  $^{\circ}$ C) and at strictly defined pH conditions (pH = 9.20), from a solution of Ni(NO<sub>3</sub>)<sub>2</sub> and Mg(NO<sub>3</sub>)<sub>2</sub> (molar ratio Mg/Ni = 0.10) using Na<sub>2</sub>CO<sub>3</sub> as the alkaline agent. The promoter role of Mg is in preventing the sintering of nickel (Ni) particles during the reduction process.

Activated domestic diatomite with a low surface area was added to the reaction mixture as a support. Upon completion of the precipitation process, the filtered and rinsed precipitate was transferred to an oven and dried at the temperature of 120-140 °C. The synthesis of the nickel catalyst supported on a natural silicate support (diatomite) is a three-stage process: the precursor synthesis, reduction, and passivation with impregnation. The preparation procedure is previously described in details (**Jovanović et al., 1998**). The calcinations step is omitted since the reduction of the original nickel basic carbonates is much easier than the reduction of NiO. The passivation and impregnation steps are not the subject of the present research.

Diatomite, used as a support, was obtained from the Baroševac – Field B ("Kolubara" coal basin, Lazarevac, Serbia). Crude diatomite material was mechanically, chemically and thermally treated in order to obtain the activated support. Specific surface area of the

support was  $11.7 \text{ m}^2 \text{ g}^{-1}$ , while the total porosity was 66.6 % with an average pore diameter of 495 nm, as determined by mercury intrusion porosimetry (**Stanković et al. 2009**).

The reduction of all of the prepared samples was studied by thermogravimetry (TG) (Linseis Thermal Analyzer L75/L81/2000 with an automatic gas control system) at different heating rates ( $\beta = 2, 5, 10$  and 20 °C min<sup>-1</sup>) in a hydrogen flow with the rate of  $\varphi$  = 50 mL min<sup>-1</sup>. Kinetic parameters for the catalyst reduced at different heating rates, can be used to explain the decrease of activity with the increasing of the heating rate.

Samples of the diatomite supported Ni catalyst with the highest value (1.15) of the  $SiO_2/Ni$  were prepared for the characterization by the reduction in a gas mixture of  $H_2/N_2$  (1/1, v/v) in an oven at 430 °C for 5 h, after the heating with a rate of 1 °C min<sup>-1</sup>. For characterization in XPS (X-ray photoelectron spectroscopy) and SEM (scanning electron microscopy), a part of the activated catalyst was passivated in mixture of 350 *ppm*  $O_2$  in nitrogen at the room temperature over the time required for reducing the catalyst's pyrophority down to the level desirable for preventing self-ignition of the reducted nickel.

The electronic states of the nickel species on the surface of the catalysts were determined by XPS (X-ray photoelectron spectroscopy) analysis on a VG Scientific ESCALAB Mk II instrument. The pressure in the analytic chamber was maintained below  $2.70 \times 10^{-6}$  Pa and the Al K<sub>a</sub> X-ray excitation source ( $h_v = 1486.6$  eV) (operating at 15 kV and 20 mA) was used. The Lab Cal 2 software is used for the data analysis. The electronic binding energy data were amended by compensating charge with C1 peak at 284.6 eV. The resolution function of the instrument has been found to have a width of 0.35 eV using the silver Fermi edge. To control charging of the samples, the charge neutralizer filament was used during all experiments. After calibration, the background from each spectrum was subtracted using a Shirley-type background to remove most of the extrinsic loss structure.

The scanning electron microscopy (SEM) analysis was performed on a JEOL Superprobe 733 microscope with a 25 kV beam. The corresponding images are represented as a secondary electron imaging (SEI). The SEM is used to investigate the morphology and the uniformity of the particle shape, as well as to determine the chemical homogeneity of the supports.

Hydrogen chemisorption at 25 °C was used to determine the nickel crystallite size and nickel surface area. The measurements were performed by the static method in a standard volumetric apparatus. The monolayer adsorption capacity was determined by extrapolation down to zero pressure with the linear regression of the equilibrium data in the range 0.1-13.3 kPa. The chemisorption measurements were carried out under the following conditions: *in situ* reduction of samples under a  $H_2/N_2$  (1/1, v/v) gas mixture stream at the heating rate of 2 °C min<sup>-1</sup>. Description of the experimental procedure and discussion concerning its adequacy were provided elsewhere (**Bartholomew, 1990; Bartholomew and Pannell, 1980; Hoang-Van et al., 1989**). The nickel surface area was

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calculated using the equation in the form  $S_{\text{Ni}} (m_{\text{Ni}}^2 g_{\text{Ni}}^{-1}) = 1.298 \times 10^{-19} \cdot N_{\text{M}}$ , where  $N_{\text{M}}$  is the number of hydrogen molecules adsorbed in the monolayer per gram of nickel. The above-stated equation was derived considering the surface occupied per atom of nickel as  $6.33 \times 10^{-2} \text{ nm}^2 \text{ atom}^{-1}$  (**Coenen and Linsen, 1970**), and the adsorption stoichiometry as 0.50 hydrogen molecules per surface nickel atom. The average size of the nickel crystallites ( $d_{s\text{Ni}}$ ) in the catalyst was calculated assuming model in which hemispherical crystallites are attached to the support with their equatorial plane. For this model, the crystallite size (defined as the cube root of the volume) was calculated from the nickel area per gram of nickel ( $S_{\text{Ni}}$ ) using the relation  $d_{s\text{Ni}}$  (nm) = 431/ $S_{\text{Ni}}$  (**Coenen and Linsen, 1970**).

### Kinetic Analysis

The temperature programmed reduction data is based on the understanding of gas-solid reaction mechanisms, which may include topochemical characteristics. In contrast to a homogeneous reaction, a topochemical reaction takes place at the interface between solid substrate and solid product. Reduction and oxidation reactions of solids may well exhibit topochemical dynamics, whereas adsorption and desorption on the catalytic surfaces are typically homogeneous reactions. Despite the differences in the intrinsic reaction mechanisms associated with various temperature-programmed techniques, the kinetic analyses largely share the same methodical basis.

The analysis of thermo-analytical data where there is a linear temperature rise ( $\beta = (T-T_o)/t$ ) has most often been based on the following rate equation:

$$\beta \frac{d\alpha}{dT} = A \exp\left(-\frac{E_a}{RT}\right) f(\alpha) \quad (1)$$

which states that the rate of the reaction is proportional to the rate coefficient with an Arrhenius temperature dependence ( $\alpha$  is the degree of conversion, and is expressed as  $\alpha = (m_0 - m_T)/(m_0 - m_f)$ , where  $m_o$ ,  $m_T$  and  $m_f$  are the initial, actual (at the current temperature, T), and final mass of the sample, respectively; A is the pre-exponential factor;  $E_a$  is the apparent activation energy; R is the gas constant and T is the absolute temperature) and  $f(\alpha)$  is the function of the degree of conversion. The function  $f(\alpha)$  originates from physico-geometric considerations and/or the law of mass action. **Table I** collects alternative reaction models  $f(\alpha)$  for solid-state kinetics and  $g(\alpha)$ , i.e. integrals of  $1/f(\alpha)$ 

(Brown et al., 1980; Vyazovkin and Wight, 1997).

The application of Eq. (1) is confined to the description of one intrinsic forward reaction or a set of consecutive reactions with one clearly rate-determining step that dominates over the whole range of conditions. The single-step model (Eq. (1)) often appears insufficient to describe the observed transformations. In addition, certain topochemical models cannot be presented in the functional form of Eq. (1) even though the intrinsic reaction is assumed to take place as a single-step forward reaction.

### Isoconversional (Model-Free) Technique

The isoconversional (model-free) methods are advocated for their ability to provide estimates of the apparent activation energy independent of reaction mechanisms (**Vyazovkin, 2000**). The essence of the isoconversional methods is the acknowledgment of the complex nature of an overall solid-phase reaction and the inexpressibility of the

apparent activation energy as a single constant (Vyazovkin, 2000). The isoconversional methods typically produce the apparent activation energy as a function of the degree of conversion. The interpretation of possible variable apparent activation energy in physicochemical terms poses serious difficulties. The overall reaction may involve one composite serial reactions scheme, where rate-controlling steps change with the reaction conditions. In addition, it may involve multiple parallel reactions with different reacting species or multiple parallel mechanisms for one reacting species. There are numerous other examples of overlapping chemical and/or physical elementary processes contributing to the overall dynamics. If the isoconversional analysis results in constant apparent activation energy, as a function of conversion, the reaction rate is presumably limited by a single rate determining step and it should be easy to discover the preexponential factor and the reaction mechanism. On the other hand, if the resulting apparent activation energy varies with conversion, then most simple mechanistic explanations are ruled out and it is useless to carry out model-fitting (Vyazovkin and Wight, 1999) with a single-step unidirectional reaction rate model. Evidently, then, the isoconversional methods do not contribute to the understanding of the underlying chemical phenomena. The utility of these methods remains an on/off measure of the simplicity/complexity of the examined reaction and the isoconversional methods only postpone the inevitable determination of a kinetic model.

In this paper, we have applied the Kissinger-Akahira-Sunose (KAS) integral isoconversional method (**Kissinger, 1957; Akahira and Sunose, 1971**). KAS method is

based on the Coats-Redfern approximation of the Arrhenius integral (**Coats and Redfern, 1964**). It was shown that:

$$\ln\left(\frac{\beta}{T^2}\right) = \ln\left[\frac{AR}{E_a g(\alpha)}\right] - \frac{E_a}{RT} \quad (2)$$

Thus, for  $\alpha = \text{const.}$ , the plot  $\ln(\beta/T^2)$  versus 1/T, obtained from thermo-analytical curves recorded at the several heating rates, should be a straight line whose slope can be used to evaluate the apparent activation energy ( $E_a$ ) as a function of the degree of conversion ( $\alpha$ ) independently of the kinetic model fitted by the appropriate reaction.

Despite the popularity of the isoconversional methods applied to thermo-analytical data, true kinetic analysis must involve establishing a kinetic model and determining all its parameters. Reporting the apparent activation energy alone does not constitute the kinetic analysis. The value is meaningless until related to a mechanistic model and other kinetic parameters.

## Invariant Kinetic Parameters (Ikp) Method

Invariant kinetic parameters (IKP) method (**Lesnikovich and Levchik, 1983, 1985**) requires several  $\alpha$ -*T* curves recorded at different heating rates. The IKP method gives values of the invariant kinetic parameters,  $E_{a,inv}$  and  $A_{inv}$ , which correspond to the true kinetic model that describes the investigated process at all heating rates (**Budrugeac et al., 2004**). It is based on the existence of the linear compensation effect (Eq.(3)) between  $E_a$  and lnA obtained for the same thermogravimetric curve applying various theoretical kinetic models:

$$\ln A = \alpha^* + \beta^* E_a \qquad (3)$$

where  $\alpha^*$  and  $\beta^*$  represent the compensation effect parameters.

These values of  $E_a$  and  $\ln A$  can be obtained by using the Coats-Redfern (CR) (Coats and Redfern, 1964) method (Eq. (4)):

$$\ln\left[\frac{g(\alpha)}{T^2}\right] = \ln\left(\frac{AR}{\beta E_a}\right) - \frac{E_a}{RT} \quad (4)$$

For each theoretical kinetic model,  $g(\alpha)$  (**Table I**), and each heating rate,  $\beta$ , from the slope and intercept of the plots  $\ln[g(\alpha)/T^2]$  versus 1/T, the apparent activation energy ( $E_a$ ) and the pre-exponential factor (A) can be found.

If the compensation effect exists, linear ln*A versus*  $E_a$  dependence should be obtained for each heating rate and should intersect in a point that corresponds to the true values of kinetic parameters, for the true kinetic model, which are called the invariant kinetic parameters,  $E_{a,inv}$  and  $A_{inv}$  (**Lesnikovich and Levchik, 1983, 1985**). The intersection is only approximate due to the fact that certain variations of the experimental condition determine the region of intersection.

In order to eliminate the influence of experimental conditions on the determination of  $E_{a,inv}$  and  $A_{inv}$ , they are determined from the slope and intercept of the supercorellation relation (**Budrugeac et al., 2004**):

$$\alpha^* = \ln A_{inv} - \beta^* E_{a,inv} (5)$$

The IKP method can be used only if the  $E_a$  does not depend on  $\alpha$ , which must be previously checked by the isoconversional methods. Then, the IKP method can be used for the numerical evaluation of  $f_{inv}(\alpha)$  function, by introducing the values of the invariant kinetic parameters  $E_{a,inv}$  and  $A_{inv}$  in Eq. (1). The shape of experimental curves  $f_{inv}(\alpha)$ versus  $\alpha$  suggests the algebraic expression of  $f(\alpha)$  function corresponding to the analyzed process.

### **Determination Of The Kinetic Model**

Once the apparent activation energy has been determined, the kinetic model which best describes a measured set of TPR data can be found. Málek (Málek, 1992) has shown that for this purpose it is useful to define two special functions, designated as  $y(\alpha)$  and  $z(\alpha)$ , which can easily be obtained by simple transformation of experimental data. Through the rearrangement of basic rate equation,  $d\alpha/dt = A\exp(-E_a/RT)f(\alpha)$ , the function  $y(\alpha)$  is defined as (Criado et al., 1989; Bulánek and Čičmanec, 2008):

$$y \, \mathbf{x} = \frac{d\alpha}{dt} \exp\left(\frac{E_a}{RT}\right) = Af \, \mathbf{x}$$
(6)

The  $y(\alpha)$  function is proportional to the  $f(\alpha)$  function. Thus, by plotting  $y(\alpha)$  dependence, normalized within [0,1] interval, the shape of the  $f(\alpha)$  function is obtained. The  $y(\alpha)$ function is, therefore, characteristic for a given kinetic  $f(\alpha)$  model and it can be used as a diagnostic tool for the kinetic model determination. The nature of the  $y(\alpha)$  function with respect to various kinetic model functions was reported previously (**Málek, 1992**). It should be stressed that the shape of the  $y(\alpha)$  function is strongly affected by the  $E_a$  value. Hence, the true apparent activation energy is decisive for a reliable determination of the kinetic model because of the correlation of the kinetic parameters (**Málek, 1992**). If the temperature rises at a constant rate  $\beta$ , the integration of Eq. (1) gives:

$$g(\alpha) = \int_{0}^{\alpha} \frac{d\alpha}{f(\alpha)} = \frac{AE_a}{\beta R} \exp(-x) \left[\frac{\pi(x)}{x}\right]$$
(7)

where  $x = E_a/RT$  represents the reduced apparent activation energy. By combining the basic rate equation and Eq. (7), an alternative kinetic equation is obtained

$$\frac{d\alpha}{dt} = \left[\frac{\beta}{T\pi(x)}\right] f(\alpha)g(\alpha) \tag{8}$$

After the rearrangement of Eq. (8), the  $z(\alpha)$  function is defined as (Málek, 1989):

$$z(\alpha) = \pi(x)\frac{d\alpha}{dt}\frac{T}{\beta} = f(\alpha)g(\alpha)$$
(9)

where  $\pi(x)$  is the approximation for the temperature (Arrhenius) integral. In this work, we used the fourth rational approximation of Senum and Yang for the  $\pi(x)$  function (Senum and Yang, 1979).

The shape of the  $y(\alpha)$  function as well as the maximum  $\alpha_p^{\infty}$  of the  $z(\alpha)$  function can be used as a guide to select the kinetic model. Both  $\alpha_m$  (the maximum of the  $y(\alpha)$  function) and  $\alpha_p^{\infty}$  (the maximum of the  $z(\alpha)$  function) parameters are especially useful in this respect. Their combination allows the determination of the most suitable kinetic model (**Málek, 1992**). The kinetic models D1, D2, D3, D4 (diffusion set of models) and RO(*n*) (reaction order set of models) have a maximum at  $\alpha_m = 0$  (**Málek, 1992**). On the other hand, the Johnson-Mehl-Avrami (JMA(*M*)) (**Johnson and Mehl, 1939; Avrami, 1939, 1940, 1941**) and the Šesták-Berggren (SB(*m*,*n*)) (**Šesták and Berggren, 1971**) models have a maximum at  $0 < \alpha_m < \alpha_p$  ( $\alpha_p$  is the conversion value at the position of peak temperature ( $T_p$ ) value). For the JMA(M) model, the relationship between the  $\alpha_m$  value and the kinetic exponent M is represented as (**Málek, 1992**):

$$\alpha_m = 1 - \exp\left(\frac{1 - M}{M}\right) \tag{10}$$

It is expressed for the SB(*m*,*n*) reaction model as (Málek, 1992):

$$\alpha_m = \frac{m}{m+n}, \quad (11)$$

where *m* and *n* represent the SB kinetic exponents (namely, in SB kinetic model expressed through the function  $\alpha^m(1-\alpha)^n$ , the involvement of reactants ( $\alpha$ ) and products (1- $\alpha$ ) is attuned to their actual chemical transience (*m*) and fertility (*n*), which is timely adjusted by this type of power-law (**Koga et al., 2013**)). An important fact is that the  $\alpha_p^{\infty}$ is invariant with respect to the kinetic exponent for the JMA(*M*) model. On the other hand, for both RO(*n*) and SB(*m*,*n*) reaction models, the parameter  $\alpha_p^{\infty}$  depends on the values of kinetic exponents. It can be pointed out, that the Šesták-Berggren (SB(*m*,*n*)) model gives the best description of the TPR data if  $\alpha_p^{\infty} \neq 0.632$  and  $\alpha_m$  ( $0,\alpha_p$ ) (**Málek**, **1989**). In accordance with the above-mentioned properties of the *y*( $\alpha$ ) and *z*( $\alpha$ ) functions, these two functions allow to check the applicability of the conventional physicogeometry based kinetic models.

#### **RESULTS AND DISCUSSION**

### Kinetic Conversion Curves Of Reduction Process

The non-isothermal conversion curves of the investigated systems are shown in **Fig. 1**. Only part of the thermo-analytical curves corresponding to the reduction process (the experimental curves are expressed in the form which is suitable for the kinetic analysis as  $\alpha = \alpha(T)$  is presented. At lower temperatures (T < 150 °C) almost constant mass loss, corresponding to the loss of sorbed water, was observed for all the samples (13-15 %). Therefore, in **Fig. 1**, and subsequently, the relative mass loss, referring to the sample mass at the beginning of the reduction process was used.

The characteristic temperature values (the initial  $(T_i)$ , peak (inflection)  $(T_p)$  and final  $(T_f)$  temperature) of the reduction process for all investigated samples are shown in the **Table II**. It can be observed from **Table II** that all temperature values increase with increasing heating rate  $(\beta)$ , which is typical for the thermally activated heterogeneous systems.

From the results given in **Fig. 1** and **Table II**, it can also be seen that the initial temperature of the reduction process depends on the heating rate  $\beta$  and not on the SiO<sub>2</sub>/Ni molar ratio (**Table II**). On the other hand, the total mass loss depends on the SiO<sub>2</sub>/Ni mole ratio and not on the heating rate (**Table III**), indicating that the complete hydrogenation of all investigated samples was achieved.

In addition, assuming that the complete hydrogenation was achieved, the composition of the precursor samples was calculated from the total mass loss. The difference in the composition of the precursor samples probably originated from the synthesis procedure and conditions, and was calculated in accordance with the following stoichiometry: NiCO<sub>3</sub>·XNi OH  $_{2}$ ·YH<sub>2</sub>O + 1 +X H<sub>2</sub>  $\rightarrow$  1 +X Ni + CO<sub>2</sub> + 1 +Y + 2X ·H<sub>2</sub>O

(12)

The precursor composition is expressed through the stoichiometric coefficients *X* and *Y* in a general formula of the form  $NiCO_3 \cdot XNi(OH)_2 \cdot YH_2O$  (**Table III**). According to these results, the stoichiometric coefficients *X* and *Y* both increase with increasing SiO<sub>2</sub>/Ni molar ratio. However, more experimental measurements would be needed for the composition of the precursor samples to be characterized with much greater precision.

### Isoconversional Analysis

Nevertheless, the obtained data of the thermo-analytical curves were analyzed in their normalized form and according to the basic thermal theory for the KAS method, the parameters  $\alpha$  and T were used to calculate the apparent activation energy ( $E_a$ ). The dependences of the apparent activation energy ( $E_a$ ) on the degree of conversion ( $\alpha$ ) for the reduction process of the silica-supported Ni-catalyst precursors with different SiO<sub>2</sub>/Ni molar ratios estimated by the KAS isoconversional method are shown in **Fig. 2**.

It can be seen that the value of the apparent activation energy for all observed mole ratio cases is practically constant in the conversion range of  $0.20 \le \alpha \le 0.80$ . In the considered conversion range, the following values of the apparent activation energy were found:  $E_a =$  $134.5 \pm 5.1$  kJ mol<sup>-1</sup> (SiO<sub>2</sub>/Ni = 0.20),  $E_a = 139.6 \pm 6.1$  kJ mol<sup>-1</sup> (SiO<sub>2</sub>/Ni = 0.80) and  $E_a$ =  $128.3 \pm 6.8$  kJ mol<sup>-1</sup> (SiO<sub>2</sub>/Ni = 1.15). It can be observed that for the highest molar ratio value (SiO<sub>2</sub>/Ni = 1.15), the apparent activation energy value (128.3 kJ mol<sup>-1</sup>) is lower in comparison with  $E_a$  for the other two molar ratio values. With the increase of the SiO<sub>2</sub>/Ni molar ratio, the value of the apparent activation energy ( $E_a$ ) decreases for about  $\Delta E_a = 8.7$  kJ mol<sup>-1</sup> (the average value). From the obtained results can be concluded that the increase of the SiO<sub>2</sub>/Ni molar ratio does not cause a drastic reduction of the  $E_a$  value. On the other hand, for the conversion ranges of  $\alpha < 0.20$  and  $\alpha > 0.80$ , small variations of the apparent activation energy ( $E_a$ ) value with  $\alpha$  can be observed (**Fig. 2**).

For the reliable determination of the kinetic model of the investigated reduction process, the accurate value of the apparent activation energy must be determined. Therefore, applying the average value of  $E_a$  within  $\alpha$  from 0.20 to 0.80, rather than within the entire range, was strongly recommended because most reactions, especially the solid-state ones, are not stable at the starting and ending periods. It can be pointed out, that as the process proceeds, reactants may undergo changes in reactivity due to the crystal defect formation, intra-crystalline strain and particle disintegration (Zhou et al., 2008). The reactivity of silica-supported nickel catalyst may not remain the constant, during the full course of reduction process. Therefore, we can expect some instability in the  $E_a$  values obtained on the basis of isoconversional approach, especially at the lower values of conversion. However, the exact reason for this kind of noticed change in  $E_a$  values at lower conversions will not be discussed at this stage. Accordingly, based on the abovementioned discussion, the average values of  $E_a$  established in  $\alpha$  range of  $0.20 \le \alpha \le 0.80$ (134.5, 139.6 and 128.3 kJ mol<sup>-1</sup>) for all observed molar ratios, represent the most accurate values of the apparent activation energy.

It can be pointed out that the kinetics of reduction has extensively been investigated. Mass transport was often found to be partly or entirely rate determining (**Szekely et al., 1973; Szekely and Evans, 1971; Yamashina and Nagamatsuya, 1966**). The observed apparent activation energy was about  $E_a = 42 \text{ kJ mol}^{-1}$ . Lower values can generally be ascribed to mass transport limitation (Szekely et al., 1973; Delmon and Roman, 1973;

Rooksby, 1943; Tien and Turkdogan, 1972). For reaction front propagation, the values of  $E_a$  in the range of 117-120 kJ mol<sup>-1</sup> were obtained (**Rooksby**, 1943). For nucleation the spread was much wider, reflecting the importance of the detailed structural perfection of the lattice and its surface: 120-188 kJ mol<sup>-1</sup> (Delmon and Roman, 1973; Grange et al., **1969**). It can be concluded that the obtained values of  $E_a$  for all observed molar ratios, belong to the third group of the  $E_a$  values (120-188 kJmol<sup>-1</sup>), which can be ascribed to the nucleation and growth processes, where the reduction under such conditions can favor the formation of Ni crystallites, modifying the surface properties of nickel. The above obtained values of  $E_a$  estimated by the KAS method for various SiO<sub>2</sub>/Ni molar ratios directly correspond to the range of  $E_a$  values belonging to reduction/oxidation processes attached to the nickel complex compounds (Wang et al., 2008). This result clearly shows us the quality and accuracy of calculated  $E_a$  values reported for this type of process, related to the silica-supported nickel catalyst. Unlike the un-supported catalysts (Janković et al., 2008; L'vov and Galwey, 2012), in the current case, the higher values of  $E_a$  were obtained. Namely, traditionally is believed that the rates of the gas-solid reactions are controlled either by the rates of molecular transport or by the chemical reaction rates on the solids (Misono, 2013). However, it should be noted that this assertion may not be valid in any case. Specifically, the fates and activities of surfaceformed radicals must be taken into account in order to successfully elucidate the kinetics and mechanisms of nickel oxide reduction (Janković et al., 2008; L'vov and Galwey, **2012**). In the case when the radicals have little difficulties to desorb, desorption of

radicals can effectively reduce the concentration of radicals on the solid surface to prolong the induction period and thus the overall reduction time of un-supported particles. However, in the presence of rigid pores structure of silica support, desorption of radicals out of the pores is greatly hindered. This can greatly contribute to slowing down the process, and demand for the higher energy to overcome the energy barrier, and thus contribute to the general increase in the *effective (apparent)* activation energy. In this case, the *ease* with which the surface-formed radicals migrate from its formation site to the silica-nickel boundary, can determines the rate of reduction the silica-supported nickel catalyst precursors. In this regard, we may note that higher values of  $E_a$  in the current case than in the case of un-supported catalyst (**Janković et al., 2008**) can be attributed to the phenomenon of Si coverage on Ni(211) surface than that on pure Ni(111).

## The Ikp Analysis

In order to apply the IKP method, the conversion range in which the value of  $E_a$  is practically constant has to be determined and that has been done through the above method. For determining the compensation effect parameters ( $\alpha^*$ ,  $\beta^*$ ), the Coats-Redfern method (**Coats and Redfern, 1964**) has been applied in the conversion range where  $E_a$  is independent of the degree of conversion ( $0.20 \le \alpha \le 0.80$ ). The values of kinetic parameters of the reduction process (ln*A* and  $E_a$ ), calculated for the considered set of reaction models (**Table I**) at different heating rates ( $\beta = 2, 5, 10$  and 20 °Cmin<sup>-1</sup>) are also provided [**Supplementary data (Tables I, II and III in the frame of supplementary data**)]. It was found that for all investigated SiO<sub>2</sub>/Ni molar ratios, the values of the kinetic parameters (ln*A* and  $E_a$ ) are strongly determined by the form of assumed  $g(\alpha)$ . As a result, almost any  $g(\alpha)$  function can satisfactorily fit data at the cost of drastic variations in the kinetic parameters, which compensate for the difference between the assumed form of  $g(\alpha)$  and the true, but in that case unknown reaction model. It was found that in the case of reduction processes for SiO<sub>2</sub>/Ni = 0.20 and SiO<sub>2</sub>/Ni = 0.80, the best (statistically equivalent) reaction models at all heating rates are F3/2 and F2 [Supplementary data (Tables I and II in the frame of supplementary data)]. On the other hand, for the reduction process on SiO<sub>2</sub>/Ni = 1.15, F1, F3/2 and F2 models represent the best (statistically equivalent) reaction models [Supplementary data (Table III in the frame of supplementary data)].

However, calculated values of the apparent activation energies,  $E_a$  [Supplementary data (see provided Tables)], are not in agreement with the values of  $E_a$  calculated by the KAS isoconversional method, which means that none of the tested reaction models corresponds to the real reaction model for the reduction process of silica-supported nickel catalyst precursors with different SiO<sub>2</sub>/Ni molar ratios. The apparent compensation effect was observed for all of the investigated systems at each of the observed heating rates, and corresponding ln*A versus*  $E_a$  dependences were analyzed (not shown).

A compensation effect of this kind is classified as a false or superficial compensation effect (**Koga and Šesták, 1991**) resulting from the parameter distortion by an inappropriate kinetic model function. The values of  $\alpha^*$  and  $\beta^*$  were calculated from the

intercepts and the slopes of the evaluated straight lines. The values of the apparent compensation effect ( $\alpha^*$ ,  $\beta^*$ ) obtained at different heating rates ( $\beta$ ), for all considered SiO<sub>2</sub>/Ni molar ratios are listed in **Table IV**. In the same table, the corresponding values of the isokinetic temperatures,  $T_{iso}$  ( $T_{iso} = 1/\beta^* R$ ) are also presented.

It can be seen from **Table IV**, that the value of  $\beta^*$  parameter decreases with increasing heating rate ( $\beta$ ), which causes the increase of the  $T_{iso}$  value for all observed SiO<sub>2</sub>/Ni molar ratios. When studying the kinetics of reactions of the same type using the correct  $f(\alpha)$  (or  $g(\alpha)$ ) function, the straight lines characterizing the kinetics of these reactions usually cross at a point with co-ordinates  $\ln k_{iso}$  (=  $\alpha^*$ ) and  $1/T_{iso}$ . This point corresponds to the isokinetic temperature  $T_{iso}$ , at which the rate constants of the reactions have the same value –  $k_{iso}$ . This means that at temperatures higher than  $T_{iso}$ , the reactions occurring at higher  $E_a$  would have higher rate (higher rate constants) than those with lower apparent activation energies (at the beginning of the reduction process for  $\alpha$  values of up to  $\alpha$  = 0.20 (**Fig. 2**)).

A linear regression from the straight line plot of  $\alpha^* = \ln A_{inv} - \beta^* E_{a,inv}$  allows the computation of the invariant kinetic parameters (the invariant pre-exponential factor,  $A_{inv}$ , and the invariant apparent activation energy,  $E_{a,inv}$ ). As  $\alpha^*$  and  $\beta^*$  are correlated by the supercorrelation relation, the corresponding values of the invariant kinetic parameters for all investigated SiO<sub>2</sub>/Ni molar ratios can be estimated and these values are presented in **Table V**.

The obtained values of the invariant activation energy,  $E_{a,inv}$  (**Table V**) are in good agreement with values of  $E_a$  calculated by the KAS isoconversional method for all observed SiO<sub>2</sub>/Ni molar ratios (the difference between these values does not exceed 10 kJ  $\text{mol}^{-1}$  ( $\Delta E_a < 10 \text{ kJ mol}^{-1}$ ) which represents a acceptable deviation). On the other hand, the values of the pre-exponential factor for solid phase reactions are expected to be in a wide range (six or seven orders of magnitude). In that case, the pre-exponential factors may vary from  $10^4$  to  $10^{18}$  min<sup>-1</sup> (Vlaev et al., 2007). The lower pre-exponential factors will often indicate a surface reaction, but if the reactions are not dependent on surface area, a low factor may indicate a 'tight' complex. The higher pre-exponential factors will usually indicate a 'loose' complex (Vlaev et al., 2007). The higher pre-exponential factors (after correction of surface area) can be also being obtained, if the complexes have free translation on the surface. Since the concentrations in the solid are not controllable in many cases, it would be convenient if the magnitude of the preexponential gave an indication of the molecularity. This appears to be true only for nonsurface-controlled reactions having low ( $< 10^9 \text{ min}^{-1}$ ) pre-exponential factors. Based on the facts above, we can conclude that the reduction process of silica-supported nickel catalyst precursors in a hydrogen atmosphere was governed by the surface-controlled reactions (>  $10^9 \text{ min}^{-1}$  (**Table V**)).

Since none of the considered reaction models (**Table I**) can fully describe thermoanalytical data, the values of the invariant kinetic parameters were used for the further numerical evaluation of  $f(\alpha)$  function. **Fig. 3** shows the curves  $f_{inv}(\alpha)$  versus  $\alpha$  obtained from Eq. (1) at different heating rates for all investigated SiO<sub>2</sub>/Ni molar ratios, using the experimental values of  $\beta(d\alpha/dT)$  and the values of the invariant kinetic parameters,  $A_{inv}$  and  $E_{a,inv}$  (**Table V**).

As shown in this figure, for all investigated systems and for all heating rates ( $\beta$ ), the curves  $f_{inv}(\alpha)$  versus  $\alpha$  exhibit the same shape, which indicates the same reduction mechanism and shows the corresponding dependences on the heating rate.

It is clear from the **Fig. 3** that the curve  $f_{inv}(\alpha)$  versus  $\alpha$  exhibits a maximum (designated by the arrow symbols (**Fig. 3**)), suggesting that the reduction of silica-supported nickel catalyst precursors in a hydrogen atmosphere occurs through the reaction mechanism which is similar to the mechanism represented by the JMA(*M*) model (for *M* > 1). An inspection of the reaction mechanism functions included in **Table I** clearly confirms that only in these cases do the *f*( $\alpha$ ) *versus*  $\alpha$  curves have a maximum at:

$$f'(\alpha) = \frac{df(\alpha)}{d\alpha} = 0.$$
 (13)

Comparing Eq. (13) with the different form of the respective  $f(\alpha)$  (**Table I**), D1, D2, D3, D4 and RO(*n*) models have a maximum at  $\alpha_{max} = 0$  (**Málek, 1992**). Bearing in mind the results presented in **Fig. 3**, we can eliminate both the diffusion and reaction order models from the further consideration. On the other hand, the JMA(*M*) and SB(*m*,*n*) kinetic models have a maximum between 0 and  $\alpha_p$  (**Málek, 1992**).

#### Accurate Determination Of Reduction Mechanism

In order to determine the most reliable kinetic model which best describes the investigated reduction process of silica-supported nickel catalyst precursors, we apply the two special functions ( $y(\alpha)$  and  $z(\alpha)$ ) introduced by Málek (**Málek, 1992**).

The dependence of  $y(\alpha)$  and  $z(\alpha)$  functions on the degree of conversion ( $\alpha$ ) for the reduction process of silica-supported nickel catalyst precursors with different SiO<sub>2</sub>/Ni molar ratios, are shown in the **Figs. 4-9**.

A concave shape for all  $y(\alpha)$  functions under analysis was identified (**Figs. 4-6**), which means that  $\alpha_m \neq 0$  and makes the use of the JMA(*M*) or SB(*m*,*n*) kinetic models possible. On the other hand, all the  $z(\alpha)$  functions exhibit well-defined maxima (**Figs. 7-9**), where for the SiO<sub>2</sub>/Ni = 1.15, we can observe a considerable dependence on the heating rate, especially for the higher heating rates (10 and 20 °C min<sup>-1</sup>) (**Fig. 9**). For the construction of  $y(\alpha)$  and  $z(\alpha)$  plots, the value of the invariant activation energy  $E_{a,inv}$  was used. The values of  $\alpha_m$ ,  $\alpha_p^{\infty}$  and  $\alpha_p$  estimated for the temperature-programmed reduction of silicasupported nickel catalyst precursors under hydrogen atmosphere are shown in **Table VI**.

It can be observed that the value of the parameter  $\alpha_m$  at all heating rates falls in the range of  $0 < \alpha_m < \alpha_p$ , while the parameter  $\alpha_p^{\infty}$  exhibits the values lower than 0.632 ( $\alpha_p^{\infty} \neq$ 0.632), which represents the characteristic  $\alpha_p^{\infty}$  value for the JMA(*M*) model. The kinetic model determination by the combination of  $y(\alpha)$  and  $z(\alpha)$  is especially suitable for the nucleation-growth processes, having the capability of distinguishing the SB models from the JMA models as complicated nucleation-growth processes, the present kinetic analysis exemplifies the successful application of the method to the non-isothermal reduction of silica-supported nickel catalyst precursors.

From the obtained results, we can conclude that the reduction of silica-supported nickel catalyst precursors in a hydrogen atmosphere, must be described with a more sophisticated kinetic model as the two-parameter Šesták-Berggren (SB(m,n)) autocatalytic model (**Šesták and Berggren, 1971**). In this respect, the two-parameter Šesták-Berggren (SB(m,n)) reaction mechanism function,  $f(\alpha)$ , can be presented as

# (Šesták and Berggren, 1971):

$$f(\alpha) = \alpha^m (1 - \alpha)^n, \quad (14)$$

in which m + n is called the order of reaction. The *m* and *n* exponents are adopted as kinetic parameters which describe the shape of measured thermo-analytical curves. This model can be understood in terms of the accommodation function (**Koga and Málek**, **1996**) expressing a deviation from the ideal case due to a more complex process. In that sense, the SB(*m*,*n*) function has an empirical character, which in turn makes it possible to describe various types of shape of thermo-analytical curves. Therefore, the SB kinetic model can be used for a quantitative description of more complex crystallization processes involving partially overlapping nucleation and growth phases. It has been shown by Málek (**Málek et al., 1989**) that the physically meaningful values of the parameter *m* should be confined in the range of 0 < m < 1. On the other hand, the parameter *n* is normally an integer, which for simple reactions ranges from 0 to 3. In contrast to the homogeneous reactions, the fractional reaction orders are often observed in the case of the heterogeneous reactions.

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Rearrangement of Eq. (11) gives the ratio of kinetic exponents, p = m/n:

$$p = \frac{\alpha_m}{1 - \alpha_m} \tag{15}$$

From Eq. (6), we obtain:

$$\ln y(\alpha) = \ln A + n \ln[\alpha^{p}(1-\alpha)].$$
(16)

The kinetic parameter *n* corresponds to the slope of the linear dependence of  $\ln y(\alpha)$ versus  $\ln[\alpha^p(1-\alpha)]$  for the conversion range of  $0.20 \le \alpha \le 0.80$ . Then, the second kinetic parameter is  $m = p \cdot n$ . From the intercept of the linear dependence presented by Eq. (16), the value of the pre-exponential factor (*A*) can be calculated.

**Table VII** lists the values of kinetic exponents (m and n), the values of reaction orders (m + n) as well as the values of A (with their average values) obtained by the procedure described above, for reduction process of silica-supported nickel catalyst precursors with different molar ratios.

It can be observed from **Table VII** that the values of SB kinetic exponents (*m* and *n*) as well as the values of reaction orders (m + n) depend on the heating rates for all considered molar ratios, while the values of *A* do not show large variations with heating rate (order of magnitude from  $10^{11}$  to  $10^{13}$ ). On the other hand, in the respect to the average values, both the values of the kinetic exponents (*m* and *n*) and reaction orders (*m* + *n*) decrease with increasing SiO<sub>2</sub>/Ni molar ratio (**Table VII**).

Based on the obtained results, we decided that  $f(\alpha) = \alpha^{0.50}(1-\alpha)^{1.42}$ ,  $f(\alpha) = \alpha^{0.48}(1-\alpha)^{1.14}$  and  $f(\alpha) = \alpha^{0.42}(1-\alpha)^{1.06}$  indeed are the best conversion functions for SiO<sub>2</sub>/Ni = 0.20, 0.80 and 1.15, respectively, because according to Šesták et al. (Šesták et al., 1973), in the case of a system that contains metallic particles, the reduction processes take place mainly along the border between the oxide and the metallic particles.

It can be pointed out that quantitative X-ray analysis of passivated reduced SiO<sub>2</sub>/Ni catalysts, showed that the amount of nickel oxide was proportional to the nickel surface area before passivation, indicating that the oxide found was present on the surface of the nickel crystallites and was formed in passivation (**Delmon, 1979**). Thus, in the non-passivated reduced catalyst no nickel oxide was present and incomplete reduction was located in the silicate foundation structure. This led to the conclusion that an oxide particle, derived from hydroxide by dehydration, once its reduction is initiated, is reduced completely (**Dalmai-Imelik et al., 1976**).

From the established Šesták-Berggren (SB) parameters (m, n), we can evaluate the reduction mechanism of silica-supported nickel catalyst precursors with different SiO<sub>2</sub>/Ni molar ratios. It can be seen from **Table VII** that for the molar ratio SiO<sub>2</sub>/Ni = 0.20, the value of *n* parameter was = 1.42 (the average value). This value means that the removal of oxygen from the surface involves hydrogen adsorption in the corresponding vicinity of the oxygen. This means that hydrogen adsorption in the vicinity of NiO<sub>x</sub> species becomes the rate-limiting reaction step. It could be considered as an inducing process that initiates the reduction process. With the increase of the SiO<sub>2</sub>/Ni molar ratio up to 1.15, the *n* 

parameter decreases and drops down to  $\approx 1.00$ . This fact could mean an acceleration of the reaction between the adsorbed hydrogen and the oxygen that belongs to the silica surface. These two changes also suggest that NiO<sub>x</sub> species could promote the reduction of SiO<sub>2</sub>. It should be expected, that hydrogen molecules were adsorbed with a moderate strength on SiO<sub>2</sub>/Ni (in comparison with MgO/Ni catalyst, for example) and were properly activated (**Liu et al., 2007; Nakagawa et al., 2012**). On the other hand, the values of *m* parameter are different comparing to the *n* values at all the SiO<sub>2</sub>/Ni molar ratios, which exhibit the decreasing trend behavior (taking into account the average value) (**Table VII**). Different values of this parameter suggest that the influence of nickel particles varies depending on the SiO<sub>2</sub>/Ni molar ratio.

According to the considerations above, when there are metallic particles on the surface of the oxide, one can be sure that the reduction process is taking place mainly around them. Yet, the possibility that some sites of the oxide surface that are far from the metallic particles are involved in the reduction process cannot be excluded. In other words, both the contracting sphere model (**Hurst et al., 1982**) and the autocatalytic model have to be applied in that case.

Furthermore, the SB kinetic exponents (m and n) should be linked with the mechanism of reduction process, but due to the empirical nature of the SB model such links are hard to find in the literature. However, some recent studies (**Munteanu et al., 2004; Munteanu and Segal, 2010**) has attempted to relate the SB(m,n) exponents to the structure of the considered material. Comparing the kinetic exponents, it can be seen that the average n

value decreases from 1.42 to 1.06, going from SiO<sub>2</sub>/Ni = 0.20 to SiO<sub>2</sub>/Ni = 1.15, respectively (**Table VII**). This behavior may indicate that the decreasing of *n* leads to an increase in crystallite sizes (**Munteanu and Segal, 2010**), with SiO<sub>2</sub>/Ni order: 0.20  $\rightarrow$ 0.80  $\rightarrow$  1.15. It has been proposed such a shift of *n* value indicates the shift of the mechanism from Langmuir-Hinshelwood to Eley-Rideal (**Munteanu and Segal, 2010**). On the other hand, the average *m* value decreases from 0.50 to 0.42, going from SiO<sub>2</sub>/Ni = 0.20 to SiO<sub>2</sub>/Ni = 1.15 (**Table VII**). This leads to a lot less (not as effective) role of the reduced phase in the overall reduction process, for the case of SiO<sub>2</sub>/Ni = 1.15 ratios.

The macroscopic process of reduction (nucleation/crystal growth) strongly depends on the concentration and distribution of nuclei in the sample. The nuclei can be of the two types, surface nuclei and the bulk nuclei. The surface nuclei are already present on the sample, while the bulk nuclei, termed as secondary nuclei, form during the course of reduction. To gain a better insight into the reaction mechanism, the proper morphological examination of the resulting reduction product and the corresponding chemisorption tests were performed.

## Sem Analysis

The SEM image of as-synthesized Ni/*D* precursors (*D* is the label for diatomite as a source of  $SiO_2$ ) is shown in **Fig. 10**.

As can be seen in **Fig. 10** that as-synthesized precursor displays well-pronounced habit. The surface consists of fragments different in shape and size. The larger ones have an effective diameter in the range 3-7  $\mu$ m. These are covered with numerous fine particles with an effective diameter from 0.5 to 3  $\mu$ m.

The SEM image of reduced Ni/D precursor (**Fig. 11**) annealed at 430  $^{\circ}$ C demonstrates visible changes in its morphology after reduction and passiviation. The sample undergoes significant packing in result of water and CO<sub>2</sub> removing. The large aggregates are covered with small particles of similar size and shape as in the case of the non-reduced precursor (**Fig. 10**).

The both SEM images (**Figs. 10** and **11**) are provided with ×2000 magnification and with a marker length equal to 15  $\mu$ m. The almost round fragment is with sizes: length  $\approx$  4.9  $\mu$ m and width  $\approx$  4.2  $\mu$ m. The effective diameter ( $d_{eff}$ ) is approximately 4.9  $\mu$ m.

It can be pointed out that the powder X-ray diffraction analysis (**Deraz et al., 2009**) shows that nickel catalyst supported on silica with different Ni loading resulted in an increase in the degree of ordering of Ni phase. In other words, it means that the increasing amounts of impregnate (Ni) facilitate the agglomeration process of Ni crystallites to the extent approximately proportional to the amount of nickel loaded. Specifically, this may be associated with the dispersion of nickel species available for the catalyzed reaction. However, the above-mentioned fact can significantly affects the kinetic developments of the entire reduction process, due to the different molar ratios of SiO<sub>2</sub>/Ni. Therefore, according to this relation, we can expect that there will be a greater or a lesser extent, to reach the segregation of Ni particles at the catalyst surface. To verify

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the validity of these claims, the additional experimental characterization of samples were performed.

#### **Chemisorption Analysis**

In chemisorption measurements, the total hydrogen uptake were determined from the zero pressure intercepts of the isotherms, thus eliminating any reversible adsorption, particularly that on the support.

The chemisorption results of Ni/D sample are listed in Table VIII.

It should be emphasized that the estimation of metal crystallite size from hydrogen uptake requires that assumptions be made regarding metal crystallite morphology. To calculate the average crystallite size, it was assumed hemispherical shape of nickel crystallites which is suitable for the supported catalysts with strong metal-support interaction. The average size of nickel crystallite of the studied catalysts deduced by static hydrogen chemisorption method was confirmed by XRDLB method (**Table VIII**). It may be noted the existence of a fairly large area of the metallic nickel surface, wherein the release of the nickel active sites for hydrogen uptake increasing (744.00  $\mu$ mol g<sub>Ni</sub><sup>-1</sup> (**Table VIII**)). From these results we can conclude that the structure sensitivity does not affect on the kinetics of hydrogen (H<sub>2</sub>) adsorption of Ni surfaces. In addition, the agreement between results obtained with the different techniques leads us to conclude that no changes in H-Ni stoichiometry took place. Because of the low crystallite size and high surface area of Ni/*D* sample, the surface nuclei initiate the reaction early. Thus, the reaction is more Langmuir-Hinshelwood type, as hydrogen dissociates rapidly on the surface nuclei and the dissociated hydrogen reacts with Ni-O active system. For this case, the important process is the reaction of the adsorbed hydrogen and the surface nuclei, probably the oxygen vacancies. This is indicated with a fairly high value of *n*, particularly at SiO<sub>2</sub>/Ni = 0.20 molar ratios (**Table VII**). For SiO<sub>2</sub>/Ni = 1.15 sample, the number of oxygen vacancies and correspondingly the surface nuclei are small. But as its reduction happens at the higher temperature, a larger number a bulk nuclei are formed. These bulk nuclei which are present inside NiO<sub>x</sub> crystals tend to grow much faster. The decline averaged values of *m* with an increasing of SiO<sub>2</sub>/Ni molar ratio (**Table VII**) indicates a decrease in Ni concentration within the reaction and also to reduce the concentration of Ni/Mg-base carbonate, which can be due to the absence of specific interactions between the active components in the reaction system.

It has been shown (**Šesták et al., 1973**) that the Šesták-Berggren (SB(m,n)) equation is equivalent to the JMA(M) equation to a first approximation. The SB(m,n) model thus includes JMA(M) model as a special case (**Málek, 1989**) and the kinetic parameters are related through the following equation  $M = 1/[1 + \ln(n) - \ln(n + m)]$ , where M represents the Avrami kinetic parameter. By applying the above-established equation, the average values of parameter M for considered SiO<sub>2</sub>/Ni molar ratios (in respect to particular value of the heating rate used) were calculated. The following average values of M were obtained: = 1.44 for SiO<sub>2</sub>/Ni = 0.20, = 1.55 for SiO<sub>2</sub>/Ni = 0.80 and = 1.50 for SiO<sub>2</sub>/Ni =

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1.15, respectively. The M value averaged over all molar ratio values amounts  $^{overall} =$ 1.50. The Johnson-Mehl-Avrami (JMA) model is derived from the first principles and hence, its kinetic parameter (M) has a physical significance. The value of M depends on the shape of the nuclei and the dimensionality of their growth, as well as on the rate of their formation (Koga et al., 2013). Based on the above estimated <sup>overall</sup> value (~ 1.50), we can conclude that the reduction process proceeds through the bulk nucleation which is the dominant mechanism, where there is a three-dimensional growth of crystals with polyhedron-like morphology (Fig. 11). Then we can conclude that the initial stage of surface crystallization occurs through a three-dimensional growth of crystals. However, based on the above-presented results, the optimal experimental combination for a highly active hydrogenation catalyst is a molar SiO<sub>2</sub>/Ni ratio of 1.15, and the low value of the heating rate with  $\beta = 2$  °C min<sup>-1</sup> (for the specified value of the heating rate, the value of SB parameter *m* is a very similar (**Table VII**)). It can be pointed out that numerous investigations have shown that metals can diffuse into a SiO<sub>2</sub> support (Scott et al., 1983). A metal silicide can be observed at the interfacial region of  $SiO_2$  yet at the temperature of 400 °C (Scott et al., 1983). Namely, in the initial stage, SiO<sub>2</sub> can be present as a layer between the metal and the silicon. Upon annealing at a sufficiently high temperature, the metal diffuses through the SiO<sub>2</sub> layer to form a silicide. The formation of the silicide gives rise to lateral non-uniformities in the layer such that the barrier to silicon diffusion is significantly lowered, i.e., silicon diffusion, no longer blocked by the SiO<sub>2</sub> layer, is facilitated. We may suppose that the higher temperatures lead to diffusion of Ni to the Si/SiO<sub>2</sub> interface, likely *via oxygen vacancies*.

It can be pointed out, that from a quantitative X-ray diffraction analysis of passivated reduced SiO<sub>2</sub>/Ni catalysts, it has been found that the amount of nickel oxide was proportional to the nickel surface area before passivation, indicating that the oxide found was present on the surface of the nickel crystallites and was formed in passivation (Coenen, 1958). Thus, in the non-passivated reduced catalyst no nickel oxide was present and incomplete reduction was located in the silicate foundation structure. In this case, we can expect that the oxide particle, derived from the solution after dehydration, once its reduction is initiated, is reduced completely. Comparing the obtained results for reduced precursors with those of non-reduced, we can note the importance of chemical interaction with the support. However, up to now, we can identify certain features in the non-reduced precursor (Fig. 10), which may be chemical – type of nickel compound- or the physical – pore structure with attendant mass transport limitations. In addition, based on the established data, we may conclude that the results do not indicate a progressive penetration of the reduction front into the catalyst particles so that the distribution of the reduced material seems to be a probably random (Fig. 11). In view of the morphological results (see SEM analyses) no indication of a spherical reaction front penetrating into the catalyst particles (below 10 µm (Homma et al., 2005)) to be presented.

#### **Xps** Analysis

The XPS is an experimental technique that is widely used in physico-chemical characterization of the surface of the catalytic material. In the case of nickel catalysts with supports, this method can determine the dispersion of metallic nickel on the basis

of peak intensities corresponding to the electronic structure of Ni  $2_{p3/2}$ , and based on the position of the peak in the XPS spectrum, we can determine the oxidation number. The catalyst precursors were characterized by XPS in order to know the difference between the nickel catalyst about the interaction of NiO with the support and the existence state of nickel on the catalyst surface.

The XPS results for Ni/*D* catalyst precursors with data analyzed transfer (ASCII data) (data in the form of square symbols ( $\Box$ ) - DAT), convert fitted XPS data (full black continuous line), metallic Ni data (symbol – 1 –) and the oxidation state Ni<sup>2+</sup> data (full blue continuous line) are shown in **Fig. 12**.

The binding energy of Ni  $2_{p3/2}$  for Ni/*D* was close to that for pure NiO (853.8 eV (**Moulder et al., 1978**)). Comparing with pure NiO (**Matsumura et al., 2000**), the Ni  $2_{p3/2}$  binding energy for Ni/*D* was 0.7 eV higher, indicating that nickel species mainly existed in the oxidative state, but interacted with SiO<sub>2</sub> to some extent. This result corresponded with the results reported by **Matsumura et al. (2000**), who indicated that the spectra for Ni  $2_{p3/2}$  could be separated into two Gaussian peaks (**Figure 12**). However, the observed anti-symmetry of the base Ni  $2_{p3/2}$  peak and its shift towards the lower energy values is a characteristic behavior for the spectra of the reduced samples, indicating the presence of nickel in the oxidation state Ni<sup>o</sup> (metallic nickel BE (bond energy) = 853.5 eV) (**Figure 12**). The XPS data for studied samples (including un-reduced and reduced-passivated precursors) are summarized in **Table IX**. The current table also shows the calculated ratio of Ni<sup>o</sup>/Ni<sup>2+</sup> surface concentrations.

We can see from **Table IX** that the nickel is presented in  $Ni^{2+}$  oxidation state on the surface of the un-reduced Ni/*D* sample. The bonding energy (BE) value may be attributed to Ni<sup>2+</sup> state in the nickel hydro-silicates on the surface of the sample. The reduction-passivation procedure provokes a strong impoverishment of the nickel on the surface of the Ni/*D* sample (**Table IX**). The XPS technique allows us to estimate the active phase dispersion on the samples surface, by determination of the active component/support atomic ratio. It is evident that the Ni/Si ratio decrease after the reduction-passivation procedure (the reduced Ni/*D* sample (**Table IX**)). A lower Ni/Si ratio may be explained by the low Ni and the higher Si surface concentrations. Obviously, the important fact that should be emphasized is that the Ni dispersion decreases. The results which are derived on the basis of XPS analysis are in full compliance with predicted kinetic analysis, and the established mechanistic explanation of the entire process.

# CONCLUSION

The kinetic process for the non-isothermal reduction of silica-supported nickel catalyst precursors with different SiO<sub>2</sub>/Ni molar ratios (SiO<sub>2</sub>/Ni = 0.20, 0.80 and 1.15) under hydrogen atmosphere was studied. The Kissinger-Akahira-Sunose (KAS) isoconversional method is used to estimate the dependency of the apparent activation energy on the degree of conversion. It was found that for all the molar ratios, the apparent activation energy is practically constant in the conversion range of  $0.20 \le \alpha \le 0.80$ . In the considered conversion range, the following values of the apparent activation energy (*E<sub>a</sub>*) were found: 134.5 kJ mol<sup>-1</sup> (SiO<sub>2</sub>/Ni = 0.20), 139.5 kJ mol<sup>-1</sup> (SiO<sub>2</sub>/Ni = 0.80) and 128.3

kJ mol<sup>-1</sup> (SiO<sub>2</sub>/Ni = 1.15). Applying the IKP method for the range where  $E_a$  is independent on  $\alpha$  led to the invariant kinetic parameters, which were used for the numerical evaluation of the differential conversion function ( $f_{inv}(\alpha)$ ) for all investigated systems. It was found that the obtained values of the invariant apparent activation energy,  $E_{a,inv}$ , are in good agreement with the values of  $E_a$  calculated by the Kissinger-Akahira-Sunose (KAS) isoconversional method for all observed SiO<sub>2</sub>/Ni molar ratios. The calculated values of  $E_a$  for various SiO<sub>2</sub>/Ni molar ratios correspond to the range of  $E_a$ values belonging to reduction/oxidation processes attached to the nickel complex compounds.

It was established that the reduction of Ni catalyst precursors with different SiO<sub>2</sub>/Ni molar ratios is a complex process and can be described by the two-parameter Šesták-Berggren (SB(*m*,*n*)) autocatalytic model ( $f(\alpha) \equiv \alpha^m (1-\alpha)^n$ ). It was found that because of the low crystallite size and high surface area of the reduced sample, the surface nuclei initiate the reaction early. Thus, it was found that the reaction is more Langmuir-Hinshelwood type, as hydrogen dissociates rapidly on the surface nuclei and the dissociated hydrogen reacts with Ni-O active system. It was concluded that the reduction process proceeds through the bulk nucleation which is the dominant mechanism, where three-dimensional growth of crystals with polyhedron-like morphology exist. By using a variety of techniques for specimens characterization, it was found that the surface Ni/Si ratio decrease after the reduction process. This has been explained by the low Ni and the higher Si surface concentrations. It has been disclosed that the Ni dispersion decreases. Knowledge of the kinetic behavior of the studied system under controlled heating mode, it allows us to carry out a controlled synthesis of Ni particles of desired sizes.

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## SUPPLEMENTARY DATA

Supplementary files supplied will be published online alongside the electronic version of the article in Journal Web Page.

Table I. (**Supplementary data**) The apparent kinetic parameters for the reduction process of Ni catalyst precursor synthesized with molar ratio of  $SiO_2/Ni = 0.20$  under hydrogen atmosphere, obtained for each kinetic function versus the heating rates

Μ	$\beta = 2$	2 °Cn	nin <sup>-1</sup>	<b>β</b> = 5	5 °Cmir	1 <sup>-1</sup>	$\beta = 10$	°Cmin	-1	$\beta = 20$	°Cmin	-1
0	Ea	ln	$R^{2a}$	Ea	lnA	$R^{2a}$	Ea	lnA	$R^{2a}$	Ea	lnA	$R^{2a}$
d	(k	A		(k			(kJ			(kJ		K
el	Jm			Jm			mol			mol		
	oľ			oľ			<sup>1</sup> )		C	1)		
	<sup>1</sup> )			<sup>1</sup> )					S			
Р	27.	2.	0.954	31.	3.92	0.9607	24.5	2.75	0.955	24.6	3.19	0.93822
1	3	10	72	3		3		2	56			
Р	39.	5.	0.961	44.	7.17	0.9658	35.8	5.49	0.962	36.1	5.88	0.94847
2	4	10	26	8		3			85			
Р	63.	10	0.966	71.	13.4	0.9700	58.3	10.7	0.968	58.9	10.9	0.95644
3	5	.8	51	7	3	2		1	52		9	
		4		$\bigcirc$								
Р	20	43	0.972	23	49.3	0.9745	193.	40.3	0.974	196.	39.9	0.96454
4	8.5	.6	04	3.3	5	4	5	1	32	0	0	
7		0										
R	13	27	0.970	15	31.5	0.9735	125.	25.6	0.973	127.	25.6	0.96273
1	6.0	.3	79	2.5	5	1	9	8	02	5	1	
		9										
R	16	33	0.985	18	38.3	0.9875	153.	31.2	0.987	155.	31.1	0.98092

2	4.9	.5	78	4.8	1	3	2	4	67	5	2	
		1										
R	17	35	0.989	19	40.6	0.9909	163.	33.1	0.991	165.	33.0	0.98547
												0.98547
3	5.7	.6	45	6.7	5	2	3	4	17	9	0	
		2										
F	19	42	0.994	22	47.6	0.9959	185.	39.1	0.996	188.	39.0	0.99247
1	8.8	.1	97	2.5	4	2	1	9	24	5	4	
		2										
F	23	51	0.999	26	57.5	0.9995	221.	47.4	0.999	226.	47.3	0.99837
									N			0.77037
3/	7.6	.1	25	5.7	1	8	8	9	73	5	3	
2		5						0				
F	28	61	0.999	31	68.5	0.9997	263.	56.8	0.999	269.	56.6	1.00000
2	1.2	.2	88	4.3	8	0	2	1	49	5	5	
		8										
F	38	84	0.995	42	93.9	0.9941	358.	78.1	0.993	368.	78.0	0.99611
												0.77011
3	1.7	.4	02	6.1	3	4	5	6	08	6	3	
		7										
Α	12	26	0.994	14	30.4	0.9957	120.	24.9	0.996	122.	25.0	0.99204
3/	9.5	.3	71	5.3	0	4	3	2	02	4	3	
2		8										
A	94.	18	0.994	10	21.6	0.9955	87.9	17.6	0.995	89.4	17.9	0.99158
2	9	.4	44	6.7	8	4		9	79		2	
		2										
		2										

Α	60.	10	0.993	68.	12.8	0.9951	55.5	10.3	0.995	56.4	10.6	0.99053
3	3	.3	84	1	3	1		0	26		7	
		1										
А	43.	6.	0.993	48.	8.29	0.9946	39.3	6.50	0.994	39.9	6.93	0.98928
4	0	14	13	8		1			64			
D	28	59	0.972	31	67.0	0.9750	261.	54.8	0.974	264.	54.0	0.96540
1	1.0	.6	63	4.1	1	3	1	1	93	5	5	
		8							Ç			
D	31	67	0.981	35	75.4	0.9836	294.	61.7	0.983	299.	60.8	0.97630
2	6.7	.3	80	4.0	1	9	8	1	85	1	7	
		0						0				
D	36	75	0.990	40	84.9	0.9913	335.	69.4	0.991	341.	68.5	0.98631
3	0.3	.8	00	2.6	4	4	9	5	67	5	6	
		9			0	5						
D	33	69	0.984	37	77.5	0.9866	308.	63.2	0.986	313.	62.4	0.98016
4	1.1	.1	98	0.1	5	8	4	7	92	1	0	
		3										

Table II. (**Supplementary data**) The apparent kinetic parameters for the reduction process of Ni catalyst precursor synthesized with molar ratio of  $SiO_2/Ni = 0.80$  under hydrogen atmosphere, obtained for each kinetic function versus the heating rates

Mod	$\beta = 2^{\circ}$	Cmin <sup>-1</sup>		$\beta = 5^{\circ}$	Cmin <sup>-1</sup>		$\beta = 10$	°Cmin	-1	$\beta = 20$	°Cmin	-1
el	Ea	lnA	$R^{2a}$	Ea	lnA	$R^{2a}$	Ea	lnA	$R^{2a}$	Ea	lnA	$R^{2a}$
	(kJm			(kJm			(kJm			(kJm		
	ol <sup>-1</sup> )			ol <sup>-1</sup> )			ol <sup>-1</sup> )	C		<b>ol</b> <sup>-1</sup> )		
P1	40.6	5.53	0.972	30.8	3.82	0.961	36.3	5.73	0.973	31.9	5.01	0.955
			67			52	$\bigcirc$		75			20
P2	57.1	9.50	0.975	44.1	7.04	0.966	51.4	9.30	0.976	45.7	8.19	0.961
			44			58			80			16
P3	90.1	17.2	0.977	70.7	13.2	0.970	81.8	16.2	0.979	73.3	14.3	0.966
		3	83		5	73		1	37		2	03
P4	288.1	62.0	0.980	230.3	48.8	0.975	263.7	56.0	0.982	239.0	49.3	0.971
		2	51		2	19		8	19		9	25
R1	189.1	39.7	0.979	150.5	31.2	0.974	172.7	36.3	0.981	156.2	32.0	0.970
÷	$\sim$	8	88		0	17		1	54		2	06
R2	227.9	48.1	0.992	182.7	37.9	0.988	206.3	43.2	0.992	190.9	38.9	0.986
X		5	05		6	56		5	53		7	17
R3	242.2	51.0	0.994	194.7	40.3	0.991	218.7	45.6	0.995	203.9	41.4	0.990
		9	79		1	96		5	02		0	04
F1	273.2	59.3	0.998	220.5	47.3	0.996	245.3	52.7	0.998	231.9	48.6	0.995

		8	44		5	82		5	41		3	69
F3/2	325.4	71.4	0.999	264.0	57.3	0.999	289.6	62.7	0.999	279.4	58.9	0.999
		5	92		3	87		5	97		5	65
F2	384.1	85.0	0.998	313.1	68.5	0.999	339.4	73.9	0.998	333.2	70.5	0.999
		1	99		4	09		0	89		9	49
F3	519.3	116.	0.988	426.3	94.2	0.991	453.2	99.3	0.989	457.3	97.3	0.992
		10	07		7	55		7	78		6	77
A3/2	179.2	38.0	0.998	144.0	30.2	0.996	160.4	34.0	0.998	151.4	31.5	0.995
		1	38		1	67		6	34		0	50
A2	132.2	27.2	0.998	105.7	21.5	0.996	118.0	24.6	0.998	111.2	22.8	0.995
		3	32		4	51		3	27		4	29
A3	85.1	16.3	0.998	67.4	12.7	0.996	75.6	15.0	0.998	70.9	14.0	0.994
		1	19		3	17		5	11		4	82
A4	61.6	10.7	0.998	48.3	8.21	0.995	54.4	10.1	0.997	50.8	9.53	0.994
		5	04	0		77		5	92			29
D1	387.1	84.1	0.980	310.1	66.3	0.975	354.7	75.7	0.982	321.9	66.6	0.971
	C	3	81			68		3	50		4	82
D2	434.9	94.5	0.988	349.7	74.7	0.984	396.4	84.4	0.989	364.6	75.2	0.981
		1	53			58		1	45		3	71
D3	493.4	106.	0.994	398.4	84.3	0.992	446.6	94.1	0.995	417.3	85.1	0.990
		48	98			34		7	24		3	51
D4	454.3	97.4	0.991	365.8	76.9	0.987	413.0	86.6	0.991	382.0	77.4	0.985
		6	12			64		3	76		9	16

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**Table III. (Supplementary data)** The apparent kinetic parameters for the reduction process of Ni catalyst precursor synthesized with molar ratio of  $SiO_2/Ni = 1.15$  under hydrogen atmosphere, obtained for each kinetic function versus the heating rates

Мо	$\beta = 2^{\circ}$	Cmin	-1	$\beta = 5^{\circ}$	Cmin	-1	$\beta = 10$	°Cmi	n <sup>-1</sup>	$\beta = 20$	°Cmi	n <sup>-1</sup>
del	Ea	lnA	$R^{2a}$	Ea	lnA	$R^{2a}$	Ea	lnA	$R^{2a}$	Ea	lnA	$R^{2a}$
	(kJ			(kJ			(kJ			(kJ		$\mathbf{D}_{\mathbf{A}}$
	mol			mol			mol			mol		
	1)			<sup>1</sup> )			<sup>1</sup> )		C	1)		
P1	30.6	3.0	0.97	33.1	4.3	0.97	26.5	3.2	0.98	26.9	3.7	0.94
		3	401		6	342		8	242		6	814
P2	43.8	6.2	0.97	47.1	7.7	0.97	38.4	6.1	0.98	39.0	6.6	0.95
		9	744		4	676		6	520		1	617
P3	70.1	12.	0.98	75.3	14.	0.97	62.2	11.	0.98	63.4	12.	0.96
		55	024	0	25	952		67	738		03	252
P4	227.	48.	0.98	244.	51.	0.98	205.	43.	0.98	209.	42.	0.96
	9	48	326	0	71	251	0	03	964	3	84	909
R1	149.	30.	0.98	159.	33.	0.98	133.	27.	0.98	136.	27.	0.96
	0	68	257	6	14	182	6	52	913	3	60	762
R2	178.	36.	0.99	192.	40.	0.99	160.	32.	0.99	165.	33.	0.98
X	0	87	316	7	08	340	4	98	703	8	42	463
R3	188.	38.	0.99	205.	42.	0.99	170.	34.	0.99	176.	35.	0.98
	7	98	553	0	49	590	3	84	850	8	43	877
F1	211.	45.	0.99	231.	49.	0.99	191.	40.	0.99	200.	41.	0.99

	6	48	868	5	65	904	5	77	990	6	73	492
F3/2	249.	54.	0.99	276.	59.	0.99	227.	48.	0.99	240.	50.	0.99
	9	46	993	0	81	970	0	81	873	6	46	944
F2	292.	64.	0.99	326.	71.	0.99	266.	57.	0.99	285.	60.	0.99
	8	47	803	2	21	677	7	78	472	9	29	966
F3	390.	87.	0.98	441.	97.	0.98	357.	78.	0.98	390.	82.	0.99
	9	31	925	7	40	573	6	25	279	5	86	319
A3/	138.	28.	0.99	151.	31.	0.99	124.	25.	0.99	130.	26.	0.99
2	1	65	861	3	75	899	6	99	990	5	84	464
A2	101.	20.	0.99	111.	22.	0.99	91.1	18.	0.99	95.5	19.	0.99
	4	14	854	2	71	895	$\odot$	50	989		30	435
A3	64.6	11.	0.99	71.1	13.	0.99	57.7	10.	0.99	60.5	11.	0.99
		49	838		52	884		86	987		62	368
A4	46.3	7.0	0.99	51.0	8.8	0.99	41.0	6.9	0.99	42.9	7.6	0.99
		6	820		3	872		3	985		7	290
D1	306.	66.	0.98	328.	70.	0.99	276.	58.	0.98	282.	57.	0.96
	8	15	359	4	14	284	5	41	988	3	95	979
D2	342.	73.	0.99	369.	78.	0.99	309.	65.	0.99	318.	65.	0.98
	9	89	026	2	76	016	7	25	496	6	16	012
D3	386.	82.	0.99	419.	88.	0.99	349.	72.	0.99	363.	73.	0.98
	3	52	575	1	58	609	9	80	859	2	33	938
D4	357.	75.	0.99	385.	81.	0.99	323.	66.	0.99	333.	66.	0.98
	3	74	246	7	00	257	0	74	650	3	85	373

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Table I. Expressions for  $f(\alpha)$  and  $g(\alpha)$  functions for some of the common mechanisms

operating in the solid state reactions

No.	Symbol	Reaction models	$f(\alpha)$	$g(\alpha)$
1	P1	Power law	$4\alpha^{3/4}$	$\alpha^{1/4}$
2	P2	Power law	3α <sup>2/3</sup>	α <sup>1/3</sup>
3	P3	Power law	2α <sup>1/2</sup>	$\alpha^{1/2}$
4	P4	Power law	2/3a <sup>-1/2</sup>	$\alpha^{3/2}$
5	R1	Zero-order (Polany-Winger equation)	1	α
6	R2	Phase-boundary controlled reaction (contracting area, i.e., bidimensional shape)	$2(1 - \alpha)^{1/2}$	$[1 - (1 - \alpha)^{1/2}]$
7	R3	Phase-boundary controlled reaction (contracting volume, i.e., tridimensional shape)	$3(1 - \alpha)^{2/3}$	$[1 - (1 - \alpha)^{1/3}]$
8	F1	First-order (Mampel)	(1 - α)	- ln (1 - α)
9	F3/2	Three-halves order	$(1 - \alpha)^{3/2}$	$2[(1 - \alpha)^{-1/2} - 1]$
10	F2	Second-order	$(1 - \alpha)^2$	$(1 - \alpha)^{-1} - 1$
11	F3	Third-order	$(1 - \alpha)^3$	$(1/2)[(1 - \alpha)^{-2} - 1]$
12	A3/2	Avrami-Erofe'ev ( $M = 1.5$ )	$(3/2)(1 - \alpha)[ - \ln (1 - \alpha)]^{1/3}$	$[-\ln(1-\alpha)]^{2/3}$
13	A2	Avrami-Erofe'ev ( $M = 2$ )	$2(1 - \alpha)[ - \ln (1 - \alpha)]^{1/2}$	$[-\ln(1-\alpha)]^{1/2}$
14	A3	Avrami-Erofe'ev $(M = 3)$	3(1 - α)[- ln (1 -	$[-\ln(1-\alpha)]^{1/3}$

			$(\alpha)]^{2/3}$	
15	A4	Avrami-Erofe'ev $(M = 4)$	4(1 - α)[- ln (1 -	$[-\ln(1-\alpha)]^{1/4}$
			$\alpha)]^{3/4}$	
16	D1	One-dimensional diffusion	1/2α	$\alpha^2$
17	D2	Two-dimensional diffusion (bidimensional	$1/[-\ln(1-\alpha)]$	$(1 - \alpha) \ln (1 - \alpha)$
		particle shape) Valensi equation	25	$\alpha$ ) + $\alpha$
18	D3	Three-dimensional diffusion (tridimensional	$3(1-\alpha)^{1/3}/2[(1-\alpha)^{1/3}/2]$	$[1 - (1 - \alpha)^{1/3}]^2$
		particle shape) Jander equation	$(\alpha)^{-1/3} - 1]$	
19	D4	Three-dimensional diffusion (tridimensional	$3/2[(1 - \alpha)^{-1/3} -$	(1 - 2α/3) - (1 -
		particle shape) Ginstling-Brounshtein	1]	$\alpha)^{2/3}$

olar rati	08								
β	SiO <sub>2</sub> /Ni	i = <b>0.20</b>		SiO <sub>2</sub> /Ni	i = <b>0.80</b>		SiO <sub>2</sub> /Ni	= 1.15	
(°Cmin <sup>-</sup>	T <sub>i</sub>	$T_p$	$T_f$	T <sub>i</sub>	$T_p$	$T_f$	T <sub>i</sub>	$T_p$	T <sub>f</sub>
<sup>1</sup> )	(°C)	(°C)	(°C)	(°C)	(°C)	(°C)	(°C)	(°C)	(°C)
2	212.96	265.35	299.73	222.59	265.05	280.58	210.59	263.09	283.58
5	227.37	274.45	306.93	233.85	274.16	302.76	222.19	277.25	296.40
10	233.15	284.53	323.51	236.64	283.36	305.00	230.86	286.96	303.17
20	255.27	304.71	359.04	255.20	301.68	330.73	254.66	303.73	340.53

Table II. The characteristic temperature values for the reduction process of Ni catalyst precursors synthesized with different SiO<sub>2</sub>/Ni molar ratios

Table III. The relative mass loss of the samples with given  $SiO_2/Ni$  molar ratios and calculated precursor composition according to the general formula of basic nickel carbonate/nickel hydroxide,  $NiCO_3 \cdot XNi(OH)_2 \cdot YH_2O$ 

Molar ratio	$\Delta m/m$	X	Y
SiO <sub>2</sub> /Ni			
0.20	0.47	0.47	2.07
0.80	0.32	0.85	2.76
1.15	0.27	1.26	3.20

Table IV. Values of the compensation effect parameters at the four different heating rates

( $\beta = 2, 5, 10$  and 20 °C min<sup>-1</sup>) for the reduction process of Ni catalyst precursors

synthesized with different SiO\_/Ni molar ratios

SiO <sub>2</sub> /Ni =	0.20			
β (°Cmin <sup>-</sup>	$\alpha^*$ (min <sup>-1</sup> )	β <sup>*</sup> (molkJ <sup>-1</sup> )	Tiso	$R^{2a}$
<sup>1</sup> )			(°C)	
2	-3.37033 ±	0.22497 ±	261.48	0.99855
	0.43847	0.00202		C
5	-2.35593 ±	0.22131 ±	270.33	0.99880
	0.43999	0.00181		
10	-1.91092 ±	0.21785 ±	278.96	0.99822
	0.43905	0.00217		
20	-1.30348 ±	0.20992 ±	299.82	0.99816
	0.43885	0.00213		
SiO <sub>2</sub> /Ni =	0.80		1	1
β (°Cmin <sup>-</sup>	$\alpha^*$ (min <sup>-1</sup> )	β <sup>*</sup> (molkJ <sup>-1</sup> )	Tiso	$R^{2a}$
<sup>1</sup> )	S		(°C)	
2	-2.99214 ±	0.22549 ±	260.25	0.99922
5	0.44236	0.00148		
5	-2.37681 ±	0.22209 ±	268.42	0.99880
	0.43765	0.00182		
10	-1.51616 ±	0.21815 ±	278.20	0.99898
	0.44081	0.00164		

20	-1.07704 ±	0.21097 ±	296.96	0.99878
	0.44079	0.00174		
SiO <sub>2</sub> /Ni =	1.15			
β (°Cmin <sup>-</sup>	$\alpha^*$ (min <sup>-1</sup> )	$\beta^*$ (molkJ <sup>-1</sup> )	Tiso	$R^{2 a}$
<sup>1</sup> )			(°C)	
2	-3.21309 ±	0.22645 ±	258.00	0.99874
	0.43985	0.00190		
5	-2.30575 ±	0.22114 ±	270.75	0.99887
	0.44276	0.00175		
10	-1.80859 ±	0.21828 ±	277.87	0.99836
	0.43798	0.00209	0	
20	-1.22053 ±	0.21026 ±	298.89	0.99837
a <b>D</b> <sup>2</sup> <b>A</b> 1 <sup>2</sup>	0.44014	0.00201		

<sup>a</sup> *R*<sup>2</sup> – Adj. R-Square;

Table V. The invariant kinetic parameters ( $E_{a,iso}$ ,  $A_{iso}$ ) values evaluated from the supercorellation relation (Eq. (5)) for the reduction process of Ni catalyst precursors synthesized with different SiO<sub>2</sub>/Ni molar ratios

SiO <sub>2</sub> /N	<b>Ni = 0.20</b>	
lnA <sub>iso</sub>	A <sub>iso</sub> (min <sup>-</sup>	E <sub>a,iso</sub>
	1)	(kJmol <sup>-1</sup> )
25.81	1.62 ×	$128.3 \pm 1.5$
	10 <sup>11</sup>	
SiO <sub>2</sub> /N	Ni = 0.80	
lnA <sub>iso</sub>	A <sub>iso</sub> (min <sup>-</sup>	E <sub>a,iso</sub>
	1)	(kJmol <sup>-1</sup> )
26.98	5.21 ×	$132.2 \pm 1.7$
	10 <sup>11</sup>	
SiO <sub>2</sub> /N	Ni = 1.15	0
InA <sub>iso</sub>	A <sub>iso</sub> (min <sup>-</sup>	E <sub>a,iso</sub>
	<sup>1</sup> )	(kJmol <sup>-1</sup> )
24.45	4.15 ×	$121.4 \pm 2.0$
	10 <sup>10</sup>	

Table VI. Values of  $\alpha_m$ ,  $\alpha_p^{\infty}$  and  $\alpha_p$  estimated for the reduction process of Ni catalyst precursors, synthesized with different SiO<sub>2</sub>/Ni molar ratios under hydrogen atmosphere at the different heating rates

В	SiO <sub>2</sub> /Ni = 0.20			$SiO_2/Ni = 0.80$			SiO <sub>2</sub> /Ni = 1.15			
(°Cmin <sup>-</sup>	am	$a_p^{\infty}$	ap	a <sub>m</sub>	$a_p^{\infty}$	$\alpha_p$	a <sub>m</sub>	$a_p^{\infty}$	$\alpha_p$	
<sup>1</sup> )										Q
2	0.254	0.52	0.47	0.32	0.56	0.56	0.28	0.58	0.54	
		6	7	7	7	7	3	5	5	
5	0.301	0.49	0.49	0.27	0.54	0.54	0.31	0.58	0.58	
		9	9	0	1	1	8	0	0	
10	0.250	0.54	0.50	0.30	0.56	0.56	0.25	0.64	0.61	
		8	2	7	2	2	2	8	3	
20	0.241	0.51	0.47	0.28	0.53	0.49	0.27	0.53	0.48	
		6	4	5	7	3	0	0	3	

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Table VII. The kinetic exponents (m, n) and reaction orders (m + n) of the conversion function  $f(\alpha) = \alpha^m (1-\alpha)^n$  (Eq. (14)) as well as the values of lnA and A, resulting from Eq. (16), for the reduction process of Ni catalyst precursors, synthesized with different SiO<sub>2</sub>/Ni molar ratios under hydrogen atmosphere, at the heating rates of  $\beta = 2, 5, 10$  and 20 °C min<sup>-1</sup>

SiO <sub>2</sub> /Ni =	0.20					
β (°Cmin <sup>-</sup>	m	n	<i>m</i> +	lnA	A (min <sup>-1</sup> )	$R^{2a}$
<sup>1</sup> )			n			S
2	0.48	1.40 ±	1.88	28.48	2.34 ×	0.99978
		0.01			10 <sup>12</sup>	
5	0.59	1.38 ±	1.97	29.09	4.30×	0.99897
		0.01			10 <sup>12</sup>	
10	0.46	1.39 ±	1.85	28.97	3.81 ×	0.99963
		0.01	$\mathbf{O}$		10 <sup>12</sup>	
20	0.48	1.50 ±	1.98	28.62	2.69 ×	0.99977
		0.01			10 <sup>12</sup>	
Average	0.50	1.42 ±	1.92	28.79	3.19 ×	-
C	$\mathbf{O}$	0.01			10 <sup>12</sup>	
$SiO_2/Ni =$	0.80					
β ( <sup>°</sup> Cmin <sup>-</sup>	m	n	<i>m</i> +	lnA	$A (\min^{-1})$	$R^{2a}$
<sup>1</sup> )			n			
2	0.48	0.98 ±	1.46	29.79	8.66 ×	0.99770
		0.01			10 <sup>12</sup>	

5	0.45	1.21 ±	1.66	30.07	1.15 ×	0.99901	
		0.01			10 <sup>13</sup>		
10	0.48	1.08 ±	1.56	30.28	1.41 ×	0.99674	
		0.01			10 <sup>13</sup>		
20	0.51	1.27 ±	1.78	29.94	1.01 ×	0.99953	
		0.01			10 <sup>13</sup>		
Average	0.48	1.14 ±	1.61	30.02	1.09 ×	- 6	
		0.01			10 <sup>13</sup>	5	
SiO <sub>2</sub> /Ni =	1.15	1		1			
$\beta$ (°Cmin <sup>-</sup>	m	n	<i>m</i> +	lnA	A (min <sup>-1</sup> )	$R^{2a}$	
<sup>1</sup> )			n		0		
2	0.42	1.06 ±	1.48	27.12	6.00 ×	0.99753	
		0.01	$\geq$		10 <sup>11</sup>		
5	0.46	0.99 ±	1.45	27.37	7.70×	0.99572	
		0.02			10 <sup>11</sup>		
10	0.29	0.85 ±	1.14	27.23	6.70 ×	0.99475	
	r.	0.01			10 <sup>11</sup>		
20	0.50	1.34 ±	1.84	27.36	7.63 ×	0.99956	
5		0.01			10 <sup>11</sup>		
Average	0.42	1.06 ±	1.48	27.27	6.97 ×	-	
		0.01			10 <sup>11</sup>		

Table VIII. Results of chemisorption analysis for Ni/*D* catalyst precursors (*D* represents the diatomite as a source of SiO<sub>2</sub>); The estimates of crystallite size from hydrogen chemisorption are also compared in the same table, with the values determined from the X-ray diffraction method line broadening (XRDLB)

Sample	H <sub>2</sub> uptake	$S_{\mathrm{Ni}}^{a} (\mathrm{m_{Ni}}^{2})$	H <sub>2</sub> chem	XRDLB	
	(µmol g <sub>Ni</sub> -	g <sub>Ni</sub> <sup>-1</sup> )	$d_{sNi}^{b}(nm)$	$d_{v_{\rm Ni}}^{\rm c}({\rm nm})$	
	1)				$\sim$
Ni/D	744.00	58.20	7.40	7.10	

<sup>a</sup> Nickel surface area;

<sup>b</sup> The average size of Ni crystallites – surface-averaged values;

<sup>c</sup> The average size of Ni crystallites – volume-averaged values;

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Table IX. The XPS data of the un-reduced and reduced-passivated precursors; BE -

binding energy in eV (electron-volts)

Sample	Surface	concent	rations (a	t. %)	Ni 2 <i>p</i> <sub>3/2</sub> BE (eV)			Nickel oxidation		
								state contrib	oution (%)	
	Ni		Ni/Si		un-red	red <sup>b</sup>		red <sup>b</sup>	2	
	un-red	red <sup>b</sup>	un-red	red <sup>b</sup>	Ni <sup>2+</sup>	Ni <sup>o</sup>	Ni <sup>2+</sup>	Ni°	Ni <sup>2+</sup>	
Ni/D	11.10	4.20	0.55	0.15	855.30	853.50	856.60	5.60	94.40	

<sup>a</sup> un-red: un-reduced sample;

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<sup>b</sup> red: reduced sample;

Figure 1. The non-isothermal conversion curves ( $\alpha = \alpha(T)$ ) for the reduction process of Ni catalyst precursors synthesized with different SiO<sub>2</sub>/Ni molar ratios (0.20, 0.80 and 1.15), recorded at different heating rates ( $\beta = 2, 5, 10$  and 20 °Cmin<sup>-1</sup> (from left to right)).

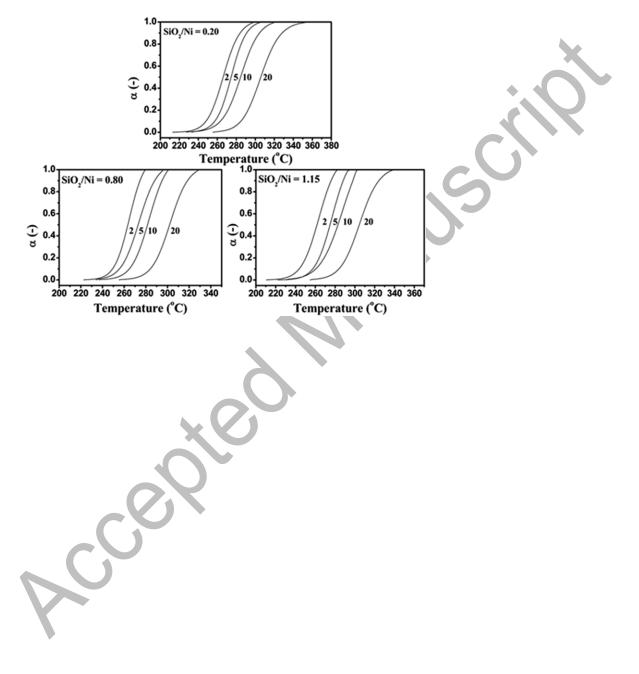


Figure 2. The dependence of the apparent activation energy  $(E_a)$  on the degree of conversion ( $\alpha$ ), determined by the Kissinger-Akahira-Sunose (KAS) method, for different

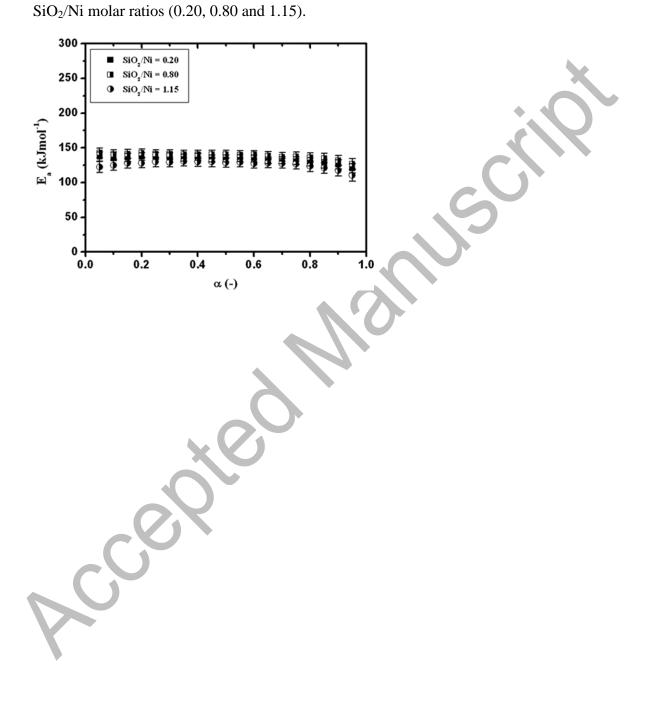


Figure 3. Dependence of  $f_{inv}(\alpha)$  versus  $\alpha$  for the reduction process of Ni catalyst precursors synthesized with different SiO<sub>2</sub>/Ni molar ratios (0.20, 0.80 and 1.15), performed at different heating rates (arrow symbol represents the position of maximum of the  $f_{inv}(\alpha)$  function at the corresponding  $\alpha$  value).

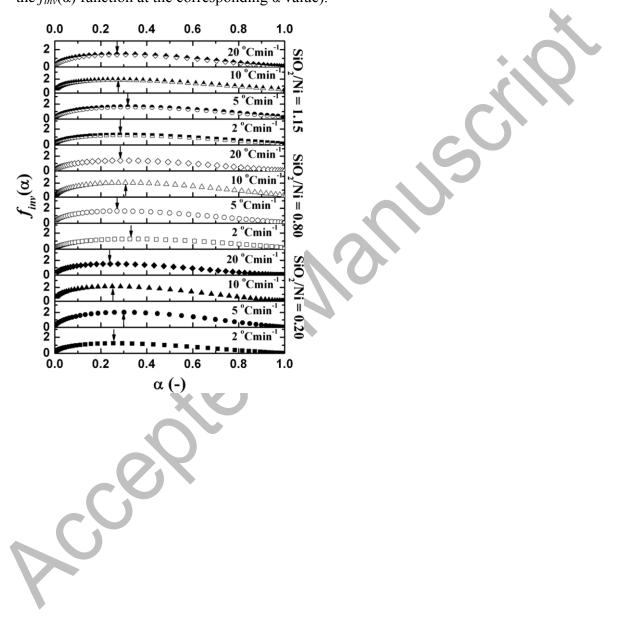


Figure 4. Normalized  $y(\alpha)$  functions obtained by the transformation of the non-isothermal thermo-analytical (TA) data, for the reduction process of Ni catalyst precursor with  $SiO_2/Ni = 0.20$  molar ratio at different heating rates.

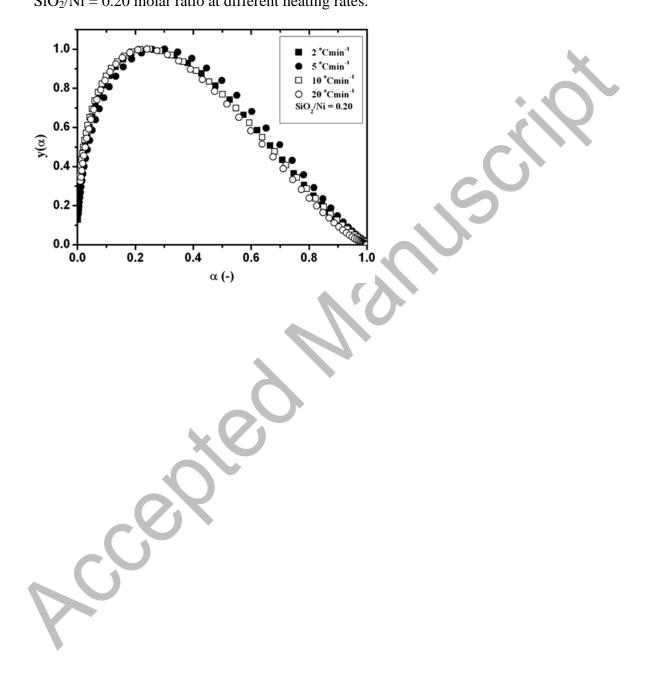


Figure 5. Normalized  $y(\alpha)$  functions obtained by the transformation of the non-isothermal thermo-analytical data, for the reduction process of Ni catalyst precursor with SiO<sub>2</sub>/Ni = 0.80 molar ratio at different heating rates.

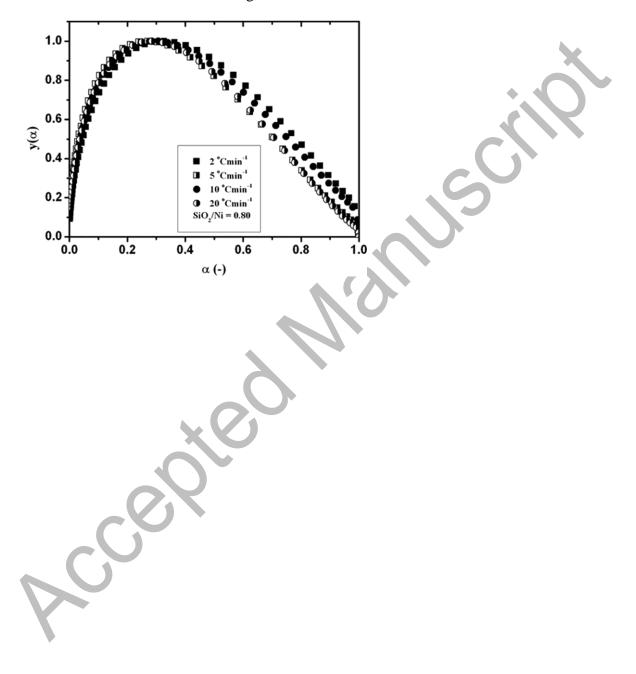


Figure 6. Normalized  $y(\alpha)$  functions obtained by the transformation of the non-isothermal thermo-analytical data, for the reduction process of Ni catalyst precursor with SiO<sub>2</sub>/Ni = 1.15 molar ratio at different heating rates.

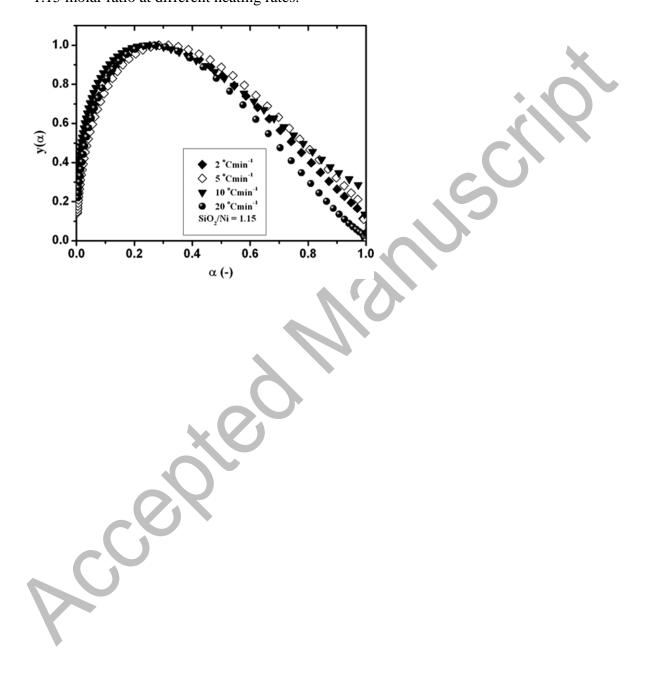


Figure 7. Normalized  $z(\alpha)$  functions obtained by the transformation of the non-isothermal thermo-analytical data, for the reduction process of Ni catalyst precursor with SiO<sub>2</sub>/Ni = 0.20 molar ratio at different heating rates.

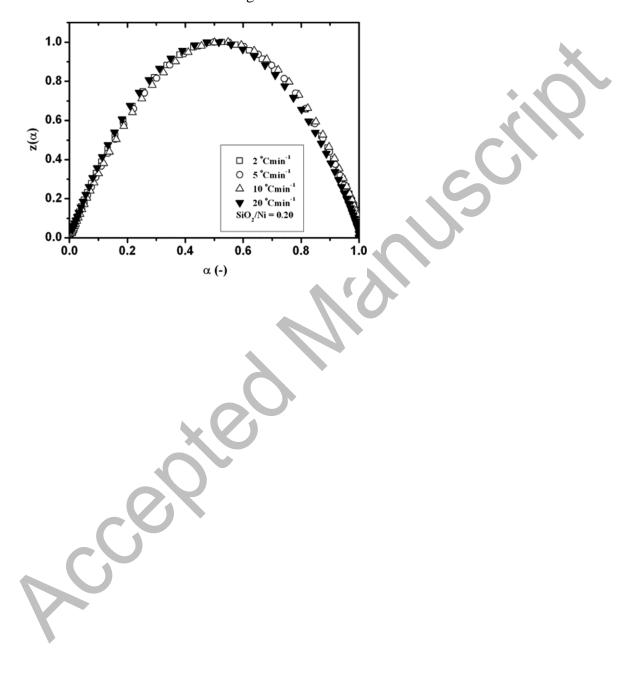


Figure 8. Normalized  $z(\alpha)$  functions obtained by the transformation of the non-isothermal thermo-analytical data, for the reduction process of Ni catalyst precursor with SiO<sub>2</sub>/Ni = 0.80 molar ratio at different heating rates.

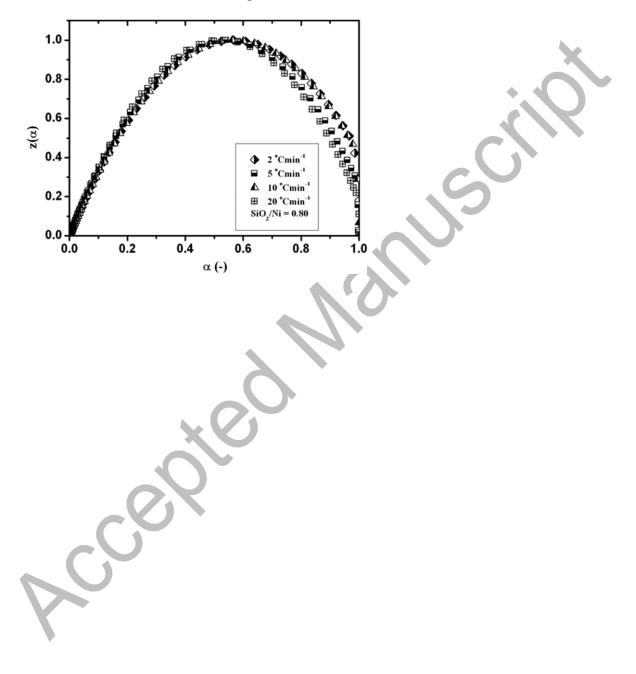


Figure 9. Normalized  $z(\alpha)$  functions obtained by the transformation of the non-isothermal thermo-analytical data, for the reduction process of Ni catalyst precursor with SiO<sub>2</sub>/Ni = 1.15 molar ratio at different heating rates.

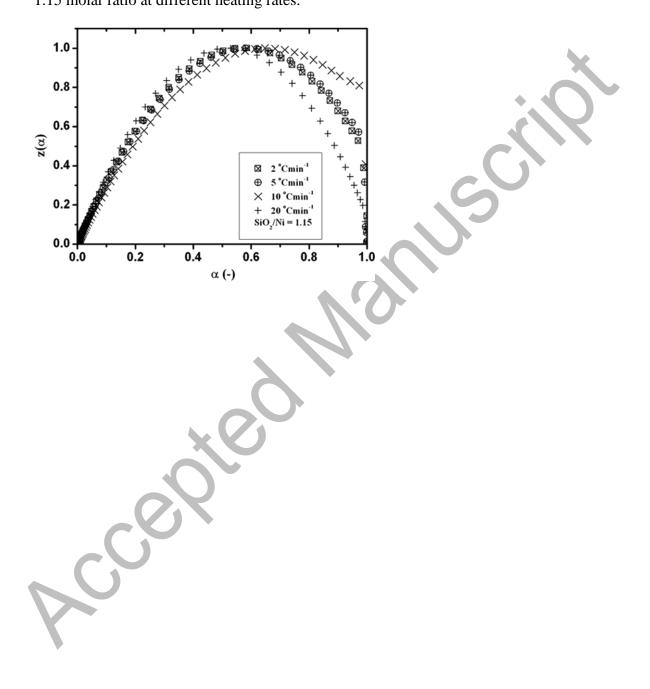
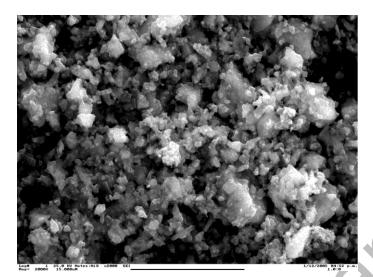


Figure 10. The SEM image with  $\times 2000$  magnification and with a marker length equal to 15 µm, for as-synthesized Ni/*D* precursors (*D* is the label for diatomite as a source of SiO<sub>2</sub>). It was taken in consideration the pattern with SiO<sub>2</sub>/Ni molar ratio equal to 1.15.



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Figure 11. The SEM image with  $\times 2000$  magnification and with a marker length equal to 15  $\mu$ m, for reduced Ni/*D* precursors (*D* is the label for diatomite as a source of SiO<sub>2</sub>) annealed at 430 °C. It was taken in consideration the pattern with SiO<sub>2</sub>/Ni molar ratio equal to 1.15.



Figure 12. The XPS imaging for Ni/*D* catalyst precursors with data analyzed transfer (ASCII data) (square symbols ( $\Box$ ) - DAT), the convert fitted XPS data (full black continuous line), metallic Ni data (symbol – 1 –) and the oxidation state Ni<sup>2+</sup> data (full blue continuous line). The energy scale is represented in electron-volts.

