

PHYSICAL CHEMISTRY 2021

15th International Conference on Fundamental and Applied Aspects of Physical Chemistry

> Proceedings Volume II

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30th Anniversary of the founding of the Society of Physical Chemists of Serbia

and

100th Anniversary of Bray-Liebhafsky reaction

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Organized by

The Society of Physical Chemists of Serbia

in co-operation with

Institute of Catalysis Bulgarian Academy of Sciences

and

Boreskov Institute of Catalysis Siberian Branch of Russian Academy of Sciences

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THERMAL PROPERTIES OF POLYURETHANE-CLAY NANOCOMPOSITES

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ABSTRACT

Two samples of poly(urethane-siloxane) nanocomposites (PUNs), based on 4,4'-diphenylmethane diisocyanate (MDI) and 1,4-butanediol (BD) as the comonomers of the hard segments (HS) and α,ω -dixydroxyethoxy propyl poly(dimethylsiloxane) (EO-PDMS) as the part of the soft segments (SS), were prepared by *in situ* polymerization. HS content of the prepared PUN samples were 10 and 20 wt%. Organomodified montmorillonite clay (Cloisite 30B[®]) was used as nanofiller (1 wt%) for poly(urethane-siloxane)s. Prepared PUNs have good thermal stability, and they represent semicrystalline polymers.

INTRODUCTION

Segmented polyurethanes (PUs) are polymers constructed of alternating flexible polyol parts that represent soft segments and rigid urethane parts which represent hard segments. By careful selection of diisocyanates, as well as polyols and chain extenders, it is possible to obtain PUs with a wide range of physical and chemical features that are quite important for a variety of applications [1]. Moreover, properties of PUs are also affected by the utilized composition of HS and SS, their chemical structure and obtained molecular weight. Although these PUs have significant number of advantages, it is still unsatisfactory when it comes to their heat resistance and thermal properties. In order to improve these properties, PUN can be prepared by using poly(dimethylsiloxane) (PDMS) as polyol and different clay nanoparticles as nanofillers [2].

EXPERIMENTAL

Two samples of PUNs based on α, ω -dixydroxyethoxy propyl poly(dimethylsiloxane) (EO-PDMS) (ABCR, $M_n = 1000$ g/mol) and organically modified montmorillonite clay (Cloisite 30B[®]), with different HS content (10 and 20 wt%, samples PUN10 and PUN20, respectively), were prepared by *in situ* two-step polyaddition reaction in THF/DMAc solution (1/1, v/v). In the first step, MDI, PDMS and delaminated clay were mixed in order to prepare –NCO-prepolymer with addition of catalyst dibutyltin dilaurate, and in the second step BD was added to the reaction mixture at 60 °C for 6 h. The PUN films were prepared by placing reaction mixtures in the Teflon[®] dishes, followed by their drying in air and vacuum oven for 24h at 50 °C and 24h at 40 °C, respectively.

The thermogravimetric analysis (TG) was carried out on a TA SDT 2960 instrument that was calibrated with Curie temperature standard materials (with well-known Curie temperatures) and melting point standards (Ni and Zn). The TG measurements were recorded under N₂ and air atmosphere, in the range from 25 to 700 °C and at a heating rate of 10 °C/min. Differential scanning calorimetry (DSC) was carried out on a Perkin-Elmer DSC6 thermal analyzer. The DSC measurements were performed under N₂ flow, using weighted samples (~5 mg) in Al pans and in the temperature range from -20 to 220 °C, at a heating and cooling rate of 10 °C/min.

RESULTS AND DISCUSSION

The thermal stability of the prepared PUNs was investigated by TG analysis under nitrogen and air atmosphere. The obtained TG and derivative TG (DTG) curves are shown in Figure 1, while characteristic degradation temperatures are listed in Table 1.

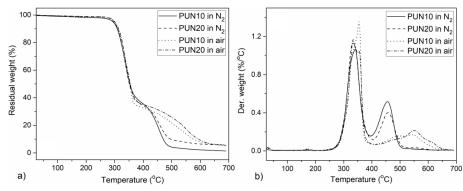


Figure 1. a) TG and b) DTG curves of the prepared PUNs in nitrogen and air atmosphere.

From the obtained TG results, thermal degradation of the PUN10 starts after 294 and 295 °C, in nitrogen and air atmosphere, respectively. $T_{10\%}$ values were 309 °C for PUN10 and 311 °C for PUN20 in nitrogen, and 311 °C for PUN10 and 314 °C for PUN20 in air atmosphere. From these results, it is concluded that PUNs exhibit slightly greater thermal stability in air than in a nitrogen atmosphere. Moreover, sample with higher content of HS exhibit greater thermal stability. According to the DTG curves (Figure 1b), thermal degradation occurs as a three- or four-step process in nitrogen and air atmosphere, respectively. In the first step, in the range from 332 to 341 °C and from 354 to 355 °C in nitrogen and air atmosphere, respectively, the urethane bonds start to decompose. In the second step, in the range from 457 to 458 °C and from 483 to 493 °C in nitrogen and air atmosphere, respectively, ether bonds inside PDMS start to break down. In the third step, the decomposition of the aromatic rings occurred in the range from 553 to 557 °C and from 547 to 549 °C in nitrogen and air atmosphere, respectively [3]. Degradation of the PUNs in the air atmosphere is more complex due to the one more additional peak-shoulder at ~ 600 °C, which originates from oxidation of the PDMS and creation of complex silicate structures.

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Sample	<i>T</i> _{5%} (°C)	<i>T</i> _{10%} (°C)	T _{50%} (°C)	<i>T</i> _{90%} (°C)	DTG _{max} (°C)
PUN10 ^{a)}	294	309	353	470	341/457/553
PUN20 ^{a)}	298	311	351	495	332/458/557
PUN10 ^{b)}		311	353	576	354/493/547/603
PUN20 ^{b)}	301	314	355	594	355/483/549/604

Table 1. TG results of the prepared PUNs in nitrogen and air atmosphere.

a) Recorded under nitrogen atmosphere

b) Recorded under air atmosphere

In order to investigate thermal properties of the prepared PUNs, differential scanning calorimetry (DSC) was performed. The obtained DSC curves are shown in Figure 2. The glass transition (T_g), melting (T_m) and crystallization (T_c) temperatures, as well as the enthalpy of melting (ΔH_m) and crystallization (ΔH_c), were determined and summarized in Table 2.

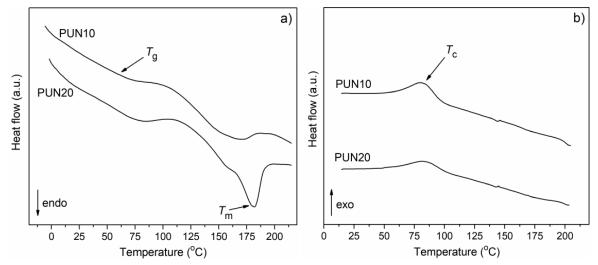


Figure 2. DSC curves of the prepared PUNs recorded during a) the second heating and b) cooling

DSC curves of PUNs have only one peak which appeared during the heating and cooling run, corresponding to the melting and crystallization, respectively, of the HS (MDI-BD). Also, PUNs have one T_g that belongs to the HS and obtained T_g values are higher than values reported for the neat copolymer, resulting from the addition of clay nanoparticles [3]. According to the results presented in Table 2, all values, except ΔH_c , increased with increasing HS content in the prepared PUNs. This was caused by better ordering of the hard domains which are formed from the longer MDI-BD units. The crystallinity degrees (X_c and X_{cHS}) were calculated based on the group contribution method [4]. Total crystallinity degrees (X_c) are lower than theoretical values, which indicates that these HS could not crystallize completely, partly because of the addition of clay nanoparticles and partly due to the incorporation of HS into the amorphous phase.

	degree of the HS (X_{cHS})) of the prepared PUNs.								
Sample	Tg (°C)	<i>T</i> _m (°C)	Tc (°C)	$\Delta H_{\rm m}$ (J/g)	$\Delta H_{\rm c}$ (J/g)	Xc (%)	Х _{сНS} (%)		
PUN10	41.4	162.9	82.8	3.54	2.46	3.88	17.81		
PUN20	54.1	181.4	84.3	10.20	1.99	11.18	30.47		

Table 2. DSC results (glass transition (T_g) , melting (T_m) and crystallization (T_c) temperatures, the enthalpy of melting ($\Delta H_{\rm m}$) and crystallization ($\Delta H_{\rm c}$), total crystallinity degree ($X_{\rm c}$) and crystallinity

CONCLUSION

Two PUN samples with different HS content, based on MDI, PDMS, BD and clay Cloisite 30B[®], were successfully prepared. TG analysis showed that these PUNs have good thermal stability, while DSC analysis showed that they are semicrystalline polymers.

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