J. Min. Metall. Sect. B-Metall. 48 (3) B (2012) 375 - 381

Journal of Mining and Metallurgy, Section B: Metallurgy

PHASE RELATIONS IN Bi - RICH PART OF THE Bi-Ga-Ni SYSTEM

D. Živković^{a,*}, D. Manasijević^a, Lj. Balanović^a, D. Minić^b, V. Ćosović^c, A. Kostov^d, Ž. Živković^a

^a University of Belgrade, Technical Faculty, Bor, Serbia

^b University of Pristina, Faculty of Technical Sciences, Kosovska Mitrovica, Serbia

^c University of Belgrade, Institute of Chemistry, Technology and Metallurgy, Belgrade, Serbia

^d Mining and Metallurgy Institute, Bor, Serbia

(Received 24 October 2012; accepted 14 November 2012)

Abstract

The results of phase relations investigation in bismuth-rich part of ternary Bi-Ga-Ni system are presented in this paper. Two sections with constant bismuth content equal to 80 and 90 at% were investigated experimentally using DSC and SEM-EDS methods and predicted thermodynamically by CALPHAD method. Calculated phase diagrams showed reasonable agreement with experimental data.

Keywords: Bi-Ga-Ni system, phase diagram, CALPHAD, DTA, SEM-EDX

1. Introduction

Gallium alloys are usually used in semiconductor processing, mirror production, for dental amalgams, low temperature solders etc. [1-6].

Some of Ga-Ni-based alloys, such as Ga-Ni-Mn, Ga-Ni-Fe or Ga-Ni-Co, are also well known due to their shape memory effect [7-9]. Also, other alloys in that group of materials, such as Ga-Ni-Me (Me=Pd, Pb, Sb, Sc, Si, Sn, Ta, Te, Ti, V, Zn, Zr), are scientifically described and examined widely [10]. Bi-Ga-Ni system has not been thermodynamically investigated, nor was its phase diagram defined yet. Constitutive binary systems - Bi-Ga [11,12], Ga-Ni [13,14] and Bi-Ni [15,16], were examined and well described in literature from thermodynamic and phase diagram point of view, while the newest thermodynamic assessment of these binaries is given in COSTMP0602 thermodynamic databases [17].

The results of experimental investigation - thermal and structural analysis, and thermodynamic prediction of phase relations in Bi-rich part of the Bi-Ga-Ni system using Calphad technique, are presented in this paper.

2. Experimental

2.1 Samples

The alloys investigated in this paper were prepared by metals - bismuth, gallium and nickel of 99.99% purity. The samples were prepared by melting

DOI:10.2298/JMMB121024047Z

in an induction furnace under protective atmosphere. The mass loss was less than 1%.

The compositions of used samples are given in Table 1.

Table 1	. Composition	of used samp	les in Bi-Ga-Ni	system –
	sections with	90%Bi (a) an	nd 80%Bi (b)	

a)						
Alloy	xBi	xGa	xNi	mass %Bi	mass %Ga	mass %Ni
E1	0,9	0,07	0,03	86,75	11,81	1,44
E2	0,9	0,04	0,06	90,00	7,00	3,00
E3	0,9	0,02	0,08	92,31	3,59	4,10
b)	b)					
Alloy	xBi	xGa	xNi	mass %Bi	mass %Ga	mass %Ni
E4	0,8	0,16	0,04	72,73	25,45	1,82
E5	0,8	0,1	0,1	78,05	17,07	4,88
E6	0,8	0,04	0,16	84,21	7,37	8,42

2.2 Techniques

Differential scanning calorimetry (DSC) and scanning electron microscopy with energy dispersive spectroscopy (SEM-EDS) were used for the experimental investigations in this paper.

Thermal analysis of chosen samples was performed using DSC method, by TA instruments SDT Q600 thermal analyzer under following

^{*} Corresponding author: dzivkovic@tf.bor.ac.rs

Dedicated to the memory of Professor Zbigniew Moser

conditions - heating rate of 5 \circ C/min, dynamic argon atmosphere, alumina crucibles, sample masses 25-30mg.

Structural analysis and equilibrium phase chemical compositions were examined by JEOL JSM-6460 scanning electron microscope equipped with Oxford Instruments energy dispersive spectrometer, under following conditions - EDS resolution up to 10 nm, acceleration voltage 0.2-30 kV and magnification up to 300000 x. No etching was applied for microscopic examination. The experimental accuracy of presented results is ± 2 at%.

3. Thermodynamic calculation of the Bi-Ga-Ni phase equilibria

Thermodynamic assessment of the Bi-Ga binary system was carried out by Girard [12]. The latest thermodynamic assessment of the Bi-Ni binary system was performed by Vassiliev et al [16] and thermodynamic assessment of the Ga-Ni binary system was done by Yuan et al. [14]. Calculated phase diagrams of the Bi-Ga, Bi-Ni and Ga-Ni binary systems are shown in Fig 1.

The phase diagram of the Bi-Ga (Fig.1a) binary system is characterized by the appearance of the liquid miscibility gap. There are no solubility in solid phases.

There are two intermediate phases in the Bi-Ni binary system (Fig.1b). The BiNi phase has very narrow homogeneity range and Bi_3Ni phase is stoichiometric compound.

The phase diagram of the Ga-Ni binary system (Fig.1c) is quite complex. It consists of the liquid solution, fcc-(Ni) solid solution with large solubility of Ga, Ga phase with no solubility of Ni and following



Figure 1. Calculated phase diagrams of binary systems - Bi-Ga [12], Bi-Ni [16] and Ga-Ni [14]

Phase and thermodynamic model	Thermodynamic parameters	Ref.
	$^{0}L^{LIQUID}_{Bi,Ga} = 8401.6 + 0.996135T$	
	$^{1}L_{Bi,Ga}^{LiQUID} = -560.9 - 2.43423T$	[12]
	$^{2}L_{Bi,Ga}^{LiQUID} = 754.8 - 0.682275T$	[12]
	${}^{3}L_{Bi,Ga}^{LIQUID} = -1162.5$	[12]
LIQUID (Bi Ga Ni)	${}^{0}L^{LIQUID}_{Bi,Ni} = -6440 + 13.288T$	[16]
	${}^{1}L_{B_{i},N_{i}}^{LiQUID} = -11315 - 1.457T$	[16]
	${}^{0}L_{Ga,Ni}^{LIQUID} = -122488.59 + 35.72T$	[14]
	${}^{1}L_{Ga,Ni}^{LiQUID} = -29685 + 14T$	[14]
	$^{2}L_{Ga,Ni}^{LIQUID} = -37051.9 + 22.10T$	[14]
FCC_A1	${}^{0}L_{Bi,NiYa}^{FCC_A1} = 20000 + 12.5T$	[16]
(Bi,Ga,Ni) ₁ (Va) ₁	${}^{0}L_{Ga}^{FCC_A1} = -130526 + 40T$	[14]
	$G_{Ga:Ni}^{BCC_B2} = -54030.75 + 16.50T + 0.5G_{Ga}^{BCC_A2} + 0.5G_{Ni}^{BCC_A}$	[14]
	$G_{GatYa}^{BCC_B2} = 5000 - 0.5T + 0.5G_{Ga}^{BCC_A2}$	[14]
	$G_{N:Ni}^{BCC_B2} = G_{Ni}^{BCC_A2}$	[14]
BCC_B2	$G_{N_i:V_a}^{BCC_B2} = -G_{Ga:N_i}^{BCC_B2} + G_{N_i:N_i}^{BCC_B2} + G_{Ga:V_a}^{BCC_B2}$	[14]
(Ga,Ni) _{0.5} (Ni,Va) _{0.5}	${}^{0}L_{Ga,Ni:Ni}^{BCC_B2} = -8724 - 2.38T$	[14]
	${}^{0}L_{Ga,Ni:Ni}^{BCC_B2} = -8724 - 2.38T$	[14]
	${}^{0}L_{Ga:N;Va}^{BCC_B2} = -35016.4 + 20.3T$	[14]
	${}^{0}L_{Ni:Ni,Va}^{BCC_B2} = -35016.4 + 20.3T$	[14]
BINI	$G_{Bi:Ni:Bi}^{BiNi} = 0 + 0.666 G_{Bi}^{SER} + 0.333 G_{Ni}^{SER}$	[16]
$(Bi)_{0.333}(Ni)_{0.333}(Bi, Ni, Va)_{0.333}$	$G_{Bi:Ni:Ya}^{BiNi} = -3550 + 0.333 G_{Bi}^{SER} + 0.333 G_{Ni}^{SER}$	[14]
	${}^{0}L^{BiNi}_{Bi:Ni:Bi,Va} = -1647 + 1.434T$	[14]
BI3NI (Bi) _{0.75} (Ni) _{0.25}	$G_{Bi;Ni}^{Bi;Ni} = -5395 + 0.75G_{Bi}^{SER} + 0.25G_{Ni}^{SER}$	[14]
	$G_{Ga:Ga:Va}^{FCC_L12} = G_{Ga}^{FCC_A1}$	[14]
	$G_{Ni:Ni:Va}^{FCC_L12} = G_{Ni}^{SER}$	[14]
	$G_{N:Ga:Va}^{FCC_L12} = -27789 + 1.64T + 0.25G_{Ni}^{SER} + 0.75G_{Ga}^{FCC_A1}$	[14]
	$G_{Ga:NEVa}^{FCC_L12} = -27789 + 1.64T + 0.75G_{Ga}^{FCC_A1} + 0.25G_{Ni}^{SER}$	[14]
	${}^{0}L^{FCC_L12}_{Ga,Ni:Ga:Va} = -55578 + 3.28T$	[14]
FCC_L12	${}^{1}L_{Ga,Ni:Ga:Va}^{FCC_L12} = 14040 - 8.01T$	[14]
$(N_1,Ga)_{0.75}(Ga,N_1)_{0.25}(Va)_1$	${}^{0}L^{FCC_L12}_{Ga,Ni:Ni:Ya} = -55578 + 3.28T$	[14]
	${}^{1}L_{Ga,Ni:Ni:Va}^{FCC_L12} = 14040 - 8.01T$	[14]
	${}^{0}L^{FCC_L12}_{Ga;Ga,Ni:Va} = 0$	[14]
	${}^{1}L_{Ga:Ga,Ni:Ya}^{FCC_L12} = 4680 - 2.67T$	[14]
	${}^{0}L_{Ni:Ga,Ni:Ya}^{FCC_L12} = 0$	[14]
	${}^{1}L_{Ni:Ga,Ni:Va}^{FCC_L12} = 4680 - 2.67T$	[14]
NI5GA3 (Ni) _{0.63} (Ga) _{0.37}	$G_{Ni:Ga}^{Ni_5Ga_3} = -37622.61 + 5.23T + 0.63G_{Ni}^{SER} + 0.37G_{Ga}^{SER}$	[14]
NI3GA2 (Ni) _{0.6} (Ga) _{0.4}	$G_{Ni:Ga}^{Ni_3Ga_2} = -39714.31 + 5.43T + 0.6G_{Ni}^{SER} + 0.4G_{Ga}^{SER}$	[14]
NI3GA4 (Ni) _{0.43} (Ga) _{0.57}	$G_{Ni:Ga}^{Ni_3Ga_4} = -47736.8 + 8.87T + 0.43G_{Ni}^{SER} + 0.57G_{Ga}^{SER}$	[14]
NI2GA3 (Ni) _{0.4} (Ga) _{0.6}	$G_{Ni:Ga}^{Ni_2Ga_3} = -47379.09T + 8.74T + 0.4G_{Ni}^{SER} + 0.6G_{Ga}^{SER}$	[14]
NIGA4 (Ni) _{0.2} (Ga) _{0.8}	$G_{N:Ga_4}^{Ni} = -24293.51 - 2.74T + 0.2G_{Ni}^{SER} + 0.8G_{Ga}^{SER}$	[14]

intermediate phases - five stoichiometric compounds Ni_5Ga_3 , Ni_3Ga_2 , Ni_3Ga_4 , Ni_2Ga_3 , and N_iGa_4 and two ordered phases - the Ni3Ga phase with the cubic L12 crystal structure (FCC_L12 thermodynamic database name) and the NiGa phase with B2-structure (BCC B2 thermodynamic database name).

Since there are no experimental information about phase equilibria and thermodynamics of investigated ternary Bi-Ga-Ni system in the literature, phase diagram of the Bi-Ga-Ni system was extrapolated using binary formalism of the CALPHAD method [18]. Thermodynamic data for the pure elements were based on the SGTE (Scientific Group Thermodata Europe) values [19]. Binary thermodynamic parameters were taken from the References [12,14,16] and shown in Table 2. Because no data are available on the solubility of Ni and Bi in (Ga) and Ga and Ni in (Bi) solid Bi and Ga were treated as a pure phases in this study.

4. Results and Discussion

The results of thermal analysis, done by DSC, are presented in Table 3. Characteristic DCS curve for the sample E4 is shown in Fig.2 as an example. Sample's solidus temperature was determined as onset temperature of the first peak on heating. Liquidus and other phase transformations temperatures were determined as maximum peak temperature on heating.

Mutual comparison between DSC results and thermodynamic prediction is shown in Fig 3.

The solidus temperature of the samples E2, E3, E5 and E6 is practically constant at 267 °C (average value). This temperature is close to the predicted temperature of ternary eutectic reaction LIQUID ->

BCC_B2 + RHOMBO_A7 + Bi3NI at 269.7 °C.

The solidus temperature of the samples E1 and E4 (227 °C average value) is considerably lower than corresponding temperature of the predicted liquid transition reaction LIQUID + NI2GA3 -> RHOMBO_A7 + NIGA4 at 253.3 °C. So, as can be notice, experimentally determined liquidus temperatures of the investigated samples are generally lower than corresponding predicted values.

Allow composition at%	Temperatures of phase transformations, °C			
Anoy composition, at/6	Solidus	Other phase transformations	Liquidus	
E1: Bi-90; Ga-7;Ni-3	228	/	896	
E2: Bi-90; Ga-4;Ni-6	267	/	905	
E3: Bi-90; Ga-2;Ni-8	268	/	875	
E4: Bi-80; Ga-16;Ni-4	226	913	995	
E5: Bi-80; Ga-10 ; Ni-10	267	/	1064	
E6: Bi-80; Ga-4 ; Ni-16	267	/	1006	

4.1 Structural analysis

Microstructure investigation of the chosen Bi-Ga-Ni alloys was performed using SEM-EDS analysis, as given in Fig. 4. The elemental compositions of the coexisting phases were determined and the obtained results are given in Table 4.

As can be seen from SEM microphotographs (Fig.3), for the samples E1 and E4 dark phase is



Figure 2. DSC heating curve for the sample E4 (Bi-80; Ga-16; Ni-4) with determined phase transformation temperatures



Figure 3. Predicted vertical sections - (a) 90 at.% Bi and (b) 80 at.% Bi - with DTA results from this work



Figure 4. Characteristic SEM microphotographs for the samples - section with 90%Bi: E1 (a), E3 (b), and section with 80%Bi: E4 (c) and E5 (d) (Note: in Fig3b black areas presents pores)

related to Ni2Ga3 intermetallic compound, gray phase corresponds to NiGa4 intermetallic compound, while light phase represents (Bi)-based solid solution (RHOMBO_A7). On the other hand, microstructures of the samples E4 and E5 consists of (Bi,Ga)-based solid solution (BCC_B2) – dark phase, Bi3Ni intermetallic compound – grey phase and (Bi)-based solid solution (RHOMBO_A7) - light phase.

According to the results of comparison between predicted and experimentally obtained phase equilibria for investigated samples, it could be seen that qualitative mutual agreement exists. However, some of the phases determined-Bi3Ni and NiGa4 include solubility of third component, which was not expected regarding to the thermodynamic prediction.

5. Conclusions

Phase relations in the Bi-rich part of the Bi-Ga-Ni system were investigated using thermodynamic prediction according to CALPHAD approach and experimentally, using SEM-EDX and DSC methods. Two vertical sections, with 90 at.% Bi and 80 at.% Bi were calculated and compared with experimental results from this work. Characteristic phase transformation temperatures of selected samples with the compositions along two investigated vertical sections were determined using DSC. Two invariant effects, at 267 °C and 227 °C, were observed on DSC heating curves. First thermal signal at 267 °C is in good agreement with predicted ternary eutectic reaction at 269.7 °C. Experimentally determined liquidus temperatures are considerably lower than predicted ones. The results of microstructure analysis confirmed the presence of the predicted phase structure at room temperature.

Acknowledgement

Financial support under project OI172037 by Ministry of Education and Science, Republic of Serbia is acknowledged. Pandat 8.1 thermodynamic software was used for the calculation presented in this work.

The authors are grateful to Prof. dr Nevenka Rajic and Mr Sanja Jevtić (University of Belgrade, Faculty of Technology and Metallurgy) and Mr Nikola Vuković (University of Belgrade, Faculty of Mining and Geology) for their help during experimental part of this work.

References

- [1] http://en.wikipedia.org/wiki/Gallium
- [2] S. Markovski, K. Micke, K. W. Richter, F. J.J. van Loo, H. Ipser, Phase relationships in the ternary Ga–Ni–Sb system, Journal of Alloys and Compounds 302 (2000) 128–136.
- [3] D. Živković, I. Katayama, L. Gomidželović, D. Manasijević, R. Novaković, Comparative thermodynamic study and phase equilibria of the Bi-Ga-Sn ternary system, International Journal of Materials Research 98 (10) (2007) 1025-1030.
- [4] C.X. Zheng, W.X., Tang, D.E. Jesson, Asymmetric coalescence of reactively wetting droplets, Applied Physics Letters, 100 (7) (2012) 071903.
- [5] J.C. Wataha, R.G. Craig, C.T. Hanks, The release of elements of dental casting alloys into cell-culture medium, Journal of Dental Research, 70 (6) (1991) 1014-1018.
- [6] H. Ipser, Basic research on lead-free soldering, J. Min. Metall. Sect. B-Metall. 43 (2) B (2007) 109-112.
- [7] R. Ducher, R. Kainuma, K. Ishida, Phase equilibria in the Ni–Fe–Ga alloy system, Journal of Alloys and Compounds, 463 (2008) 213–219.
- [8] K. Oikawa, T. Ota, Y. Imano, T. Omori, R. Kainuma, K.

 Table 4. Comparison between predicted (CALPHAD) and experimentally determined (EDX) phases for investigated sections

Sample	Predicted phases	Experimentally determined phases (EDX)	at% Ni	at% Ga	at% Bi
	RHOMBO_A7	RHOMBO_A7		0.99	99.01
E1	NI2GA3	NI2GA3		61.48	0
	NIGA4	NIGA4	7.05	84.53	8.43
	RHOMBO_A7	RHOMBO_A7	0.75	0	99.25
E3	BCC_B2	BCC_B2	55.48	44.52	0
	BI3NI	BI3NI	16.845	12.085	71.07
E4	RHOMBO_A7	RHOMBO_A7	0	0	100
	NIGA4	NIGA4	13.92	85.12	0.96
	NI2GA3	NI2GA3	38.04	61.69	0.28
E6	RHOMBO_A7	RHOMBO_A7	0.25	0	99.75
	NI2GA3	NI2GA3	38.26	61.74	0
	BCC_B2	BCC_B2	55.38	44.62	0

Ishida, Phase Equilibria and Phase Transformation of Co-Ni-Ga Ferromagnetic Shape Memory Alloy System, Journal of Phase Equilibria and Diffusion, 27 (1) (2006) 75-82.

- [9] V.D. Buchelnikov, S.V. Taskaev, M.A. Zagrebin, D.I. Ermakov, V.V. Koledov, V.G. Shavrov, T. Takagi, The phase diagrams of Ni–Mn–Ga alloys in the magnetic field, Journal of Magnetism and Magnetic Materials 313 (2007) 312–316.
- [10] http://www1.asminternational.org/asmenterprise/apd/ BrowseAPD.aspx
- [11] F. Sommer, Demixing Liquid Alloys, Zeitschrift fuer Metallkunde, 87 (1996) 865-873.
- [12] C. Girard: Thesis, University of Provence, Marseille 1985.
- [13] H. Okamoto, Ga-Ni (Gallium-Nickel), Journal of Phase Equilibria and Diffusion, 31 (2010) 575-576.
- [14] W.X. Yuan, Z. Qiao, H. Ipser, G. Eriksson, J Phase Equilib. 2004, 25(1), 68-74.
- [15] S.K. Seo, M.G. Cho, H.M. Lee, Thermodynamic assessment of the Ni-Bi binary system and phase equilibria of the Sn-Bi-Ni ternary system, Journal of Electronic Materials, 36 (11) (2007) 1536-1544.
- [16] G.P. Vassilev, V. Gandova, P. Docheva, Cryst. Res. Technol. 44 (2009) 25.
- [17] A.T. Dinsdale, A. Kroupa, J. Vizdal, J. Vrestal, A. Watson, A. Zemanova, COST Action MP0602, Version 1.0, Thermodynamic Database, Brno, Czech Republic, 2009.
- [18] H.L.Lukas, S.G.Fries, B.Sundman, Computational Thermodynamics: CALPHAD method, Cambridge University Press, Cambridge, UK 2007.
- [19] http://www.sgte.org